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WADC TECHNICAL REPORT 56-585 Part II

# ON THE MECHANICAL PROPERTIES OF AIRCRAFT STRUCTURAL METALLIC MATERIALS

Part II. STRESSED EXPOSURE OF 7075-T6

C. D. Brownfield
D. M. Badger

Northrop Corporation

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WRIGHT AIR DEVELOPMENT DIVISION

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SEPTEMBER 1960

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WRIGHT AIR DEVELOPMENT DIVISION
AIR RESEARCH AND DEVELOPMENT COMMAND
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This report covers work conducted from May 1958 to September 1959.

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#### ABSTRACT

A study has been made on the problem of predicting strength of a har ened metal alloy after subjection to variable thermal and stress environments severe enough to cause permanent loss of properties. Methods have been developed for predicting tensile ultimate, tensile yield, and compressive yield strengths of 7075-T6 aluminum alloy after single or multiple exposures to various conditions of temperature and stress. An analytical expression suitable for automatic computing machine use has also been developed.

The results of tensile and compressive tests on alclad 7075-T6 aluminum alloy showed that stresses large enough to produce inelastic creep strain during thermal exposure cause reduction in residual strength after exposure. The test results have been used to establish the usefulness of the Larson-Miller exposure parameter for correlating residual strength after simple and complex, stressed and unstressed exposures.

#### PUBLICATION REVIEW

This report has been reviewed and is approved.

FOR THE COMMANDER:

W. J. TRAPP

Chief, Strength and Dynamics Branch Metals and Ceremics Laboratory

Meterials Centrul

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#### NOMENCLATURE

	t a Paul	Units
D	$\frac{ F_{f}-F_{1} }{ F_{f}-F_{1} } _{T_{f}} \text{, strength deterioration factor describing } r_{of}$ in terms of its relationship to the strength interval between $F_{f}$ and $F_{1}$ at any temperature $T_{f}$ .	
$\mathbf{F}_{\mathbf{f}}$	Strength ( $F_{tu}$ , $F_{ty}$ , or $F_{cy}$ ) at any temperature $T_f$ with no previous exposure to temperature.	psi
lef	Strength ( $F_{tu}$ , $F_{ty}$ , or $F_{cy}$ ) at any temperature $T_f$ after exposure to $T_e$ .	p <b>si</b>
$F_1$	Strength ( $F_{tu}$ , $F_{ty}$ , or $F_{cy}$ ) at the given temperature after exposure to a reference exposure condition.	p <b>si</b>
Ftu	Tensile ultimate strength	psi
$F_{ t t y}$	Tensile yield strength	pei
F <sub>e<b>y</b></sub>	Compressive yield strength	pei
R <sub>T</sub>	Strength reduction factor combining exposure temperature effects and test temperature effects. Ratio of strength-at-temperature-after-exposure to original-room-temperature-strength.	
Rę	Strength reduction factor for effect of stress during exposure. Ratio of strength-after-stressed-exposure to strength-after-unstressed-exposure at the same value of 9.	
R.T.	(Subscript) room temperature	
T <sub>e</sub>	Temperature of exposure	°F or °R
Tf	Temperature of strength test or of design condition after exposure.	op
t	Time	hours
•	Larson-Hiller exposure parameter expressing equivalent combinations of exposure temperature and exposure time. $\theta$ = T (C + log $_{10}$ t)	(T in °k)
<b>e</b> <sub>17</sub>	$T(17 + \log_{10} t)$ , the specific variation of 9 used in this report.	(T in Ok)
••	$T(20 + 1.46 \log_{10} t)$ , the variation used for 7075-T6 in Part I.	(T in OR)
•	Inelastic strain remaining after creep exposure.	*

#### INTRODUCTION

Modern flight vehicles are sometimes subjected to such severe thermal environments that the materials of which they are constructed suffer permanent loss in strength. Typically, these materials are hardened metal alloys which progressively lose strength during periods of exposure to temperatures at which the hardening mechanism is unstable. If stress is also present during the thermal exposure, as is often the case, this can have a further damaging effect on strength. Efficient design for such severe temperatures therefore requires a knowledge of how to predict the residual strength of structural materials after exposure to various histories of temperature and stress.

In the study reported here, empirical methods were established for predicting residual strength of aluminum alloy 7075-T6 in tension and in compression after such exposures. These methods are extensions of the approaches established by Fortney and Avery in Part I on aluminum alloys 7075-T6 and 2024-T3. The choice of 7075-T6 for the present study was made recognizing that this material is finding considerable use in moderate temperature applications and some use for limited times in higher temperature applications. Cost factors often make an aluminum alloy the best choice of materials for severe thermal and stress environments, providing accurate strength data are available. The methods of analysis developed in this study allow the use of the subject material to its practical limit in high temperature design.

In the most general form, these methods are based on curves which attempt to describe the characteristic deterioration or retrogression of strength with exposure, from fully hardened values toward annealed values. For unstressed exposures of 7075-T6 it was found that a single curve could be used to adequately describe characteristic deterioration of all strength properties considered (tensile yield and ultimate, and compressive yield) throughout most of the exposure range. For stressed exposures the same approach was used with modifications. Analytical methods for design use based on these curves are given in a special section presented early in this report.

For many other hardened materials it is considered likely that similar characteristic curves can be generated by applying the methods developed in this study. For materials which respond to thermal exposure in a more complex fashion than 7075-T6 or which have a markedly different response to stress during exposure, these methods would have to be modified, possibly toward a less general type of final curve.

The empirical approach used to define the characteristics of 7075-T6 in this study was as follows: .063 inch Alclad sheet coupons were subjected to a variety of exposure conditions and then tested at selected temperatures to determine residual short-time strength. Stressed exposures were studied by applying tensile stress during exposure with the amount of stress chosen to produce up to 1 percent inelastic strain — about the practical limit for design based on short-time strength after exposure.

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The scope of the single exposure was:

250-600°F Temp. range 1-100 hr Time range

0-50,000 psi (tension)\* Stress range

The scope of multiple exposures was:

275-400°F Temp. range 3.5-340 hr Time range

to 1 percent accumulated inelastic strain Stress range

The variety of multiple exposures included:

1-step, 2-step, 3-step, 4-ster, and 10-step sequences Increasing temperature sequences Decreasing temperature sequences Mixed sequences Stressed sequences to various strain levels Stressed sequences with zero-stress soaking periods at start or end

The scope of final test conditions after exposure included:

Several mechanical properties:  $F_{tu}$ ,  $F_{ty}$ ,  $F_{cy}$ , percent elongation, hardness Several test temperatures: R.T., 200, 300, 400°F

.01 in./in./min. to yield One strain rate:

.01-.05 in./in./min.

The presentation of information in this report is oriented from the standpoint of design use. Major results are presented first followed by substantiation and detailed descriptions. Illustrative material is grouped together at the end of the report for convenience in following references frequently repeated throughout the report. This places the graphs of final results (Figure 17 through 20) which are most useful for design applications just inside the back cover where they can be found quickly.

\*Note: A spot-check of the effect of compressive exposure stress was also included.

Empirical methods for predicting residual strength of two aircraft structural materials after exposures to temperature alone, without stress have been developed in the first phase of this work, reported by Fortney and Avery in WADC Technical Report 56-585, Part I. The subject materials were 2024-T3 and alriad 7075-T6 aluminum alloys, and strengths in tension only were considered. The use of the larson-Miller time-temperautre parameter for predicting the total effect of a sequence of differing unstressed exposures was successfully demonstrated in the aforementioned report. Extension of these methods to the case of s'ressed exposures and to design mechanical properties other than yield and ultimate strengths in tension was the purpose of the study reported here. Other studies, on alloys other than 7075-T6, have considered effects of prior creep on subsequent strength for creep strains ranging above about 0.5 percent (References 4, 5, and 6). None of these referenced studies were primarily concerned with development of methods for accounting for prior creep exposure effects in practical design cases.

#### COMCLUSIONS

A number of valuable conclusions can be drawn from the results obtained in the study. The limitations of these conclusions are cited in the text.

- 1. Stress applied during severe thermal exposure can significantly accelerate deterioration of strength in 7075-T6 (retrogression of strength level toward the level of the annealed state).
- 2. The degree of strength reduction due to exposure stress is associated with the amount of inelastic strain accumulated during exposure. It is dependent upon the total degree of thermal exposure but is affected only slightly by manner of exposure accumulation for the cases investigated.
- 3. The Larson-Miller exposure parameter can be used to define equivalent single exposures for determining residual tensile ulcimate, tensile yield, and compressive yield strengths after single or multiple stressed or unstressed thermal exposures.
- 4. The general strength deterioration characteristics of 7075-T6 can be adequately represented on a single curve. Tensile ultimate, tensile yield, and compressive yield strengths and all final test temperatures are reflected in this curve. For stressed exposures, separate curves are required for different degrees of straining during exposure.

Based on these findings, analytical methods for determining allowable strengths in practical design situations have been developed. These are presented in the following section, together with limitations on their usage. Substantiation of both the conclusions and the methods can be found under experimental results.

In practical design situations it is sometimes necessary to predict allowable short-time strength after complex histories of temperatures severe enough to cause partial annealing and of stress severe enough to cause creep. "Short-time" or "static" failures as opposed to "creep" failures are a likelihood for environments in which there is considerable variation in the severity of stress-temperature cycles, and in which the severe stresses are of short duration. Such environments may or may not produce amounts of creep significant from the standpoint of design.

Analytical procedures for assessment of strength under such conditions are presented here. In the approach used, the failure is artifically separated from the spectrum, and thereby the effects of the prior history on the allowable strength remaining may be defined. The prior history is simplified by computing an equivalent exposure value for the temperature spectrum alone and by accounting for the effects of creep separately.

To find equivalent thermal exposures, the Larson-Miller exposure parameter, 0, is used:

$$\Theta = T(C + \log t) \tag{1}$$

where T is temperature in degrees Rankine, t is time in hours, and C is a constant. For the 7075-T6 data used in this report, the best value of C has been found to be 17, and the parameter becomes:

$$\theta_{17} = T(17 + \log t) \tag{2}$$

Given an actual spectrum of temperature exposures, the first step in computing the equivalent total exposure for 7075-T6 is to reduce each increment of exposure in the spectrum to a value of  $\theta_{17}$  using equation (2). Next, a single reference temperature,  $T_{\rm ref}$  is chosen arbitrarily, preferably within the temperature range of the spectrum. The equivalent time at this temperature is computed using equation (2) for each of the previously determined values of  $\theta_{17}$ . Adding the results gives the total equivalent time at  $T_{\rm ref}$  for the entire spectrum, as shown below:

Given Spectrum	<u> </u>	Find Time at Tref
T <sub>1</sub> , t <sub>1</sub>	$(e_{17})_1$	Equiv t <sub>1</sub>
<sup>7</sup> 2, <sup>t</sup> 2	$(9_{17})_2$	Equiv t <sub>2</sub>
T3, t3	( <b>9</b> <sub>17</sub> ) <sub>3</sub>	Equiv t3
•	•	•
•	•	•
•	•	•
		Total Equiv. Time

The equivalent total thermal exposure,  $\theta_{17}$  is now computed directly from equation (2) using the total equivalent time and the reference temperature.

The evaluation of a spectrum can be considerably simplified by using Figure 21. With this diagram the only calculation required for the above example is addition of the equivalent times at  $T_{\rm ref}$ .

Once the equivalent total exposure is known, the assessment of its effect (if any) on strength can be made. It is useful to separately identify the major sources of strength reduction encountered. Strength is reduced as a result of: (1) duration of exposure to a sufficiently high temperature and (2) temperature at the time of failure. Combining these two effects, the residual strength after temperature exposure and at a given test temperature can be expressed as a function of original room temperature strength, as follows:

$$F_{ef} = R_T F_{R,T}. \tag{3}$$

where  $F_{\rm ef}$  is residual strength ( $F_{\rm tu}$ ,  $F_{\rm ty}$ , etc.),  $R_T$  is a strength reduction factor combining exposure temperature effects and test temperature effects and  $F_{R-T}$  is the original room temperature strength.

Stress during exposure, especially if high enough to cause inelastic strain, is another factor which may affect residual or final strength. It will be seen that stress did have a significant effect on the 7075-T6 material tested in this work. Including this factor, equation (3) above becomes,

$$F_{ef} = R_T R_{\epsilon} F_{R,T}. \tag{4}$$

where  $R_{\varepsilon}$  is a strength reduction factor for the effect of strain accumulated during stressed exposures.

Values of RT, R $_{\rm c}$ , and RTR $_{\rm c}$  obtained for 7075-T6 are shown in Figures 16 and 17 plotted against exposure value  $\theta_{17}$ . These were obtained from a program of single exposure tests and checked at key points by program of sequential exposure tests. Ftu, Fty or Fcy values may be determined for a variety of exposure conditions and test temperatures using these data and equation (4). The FR.T. value used is the corresponding allowable room temperature Ftu, Fty or Fcy (see Reference 7). Limitations on the use of these data are discussed at the end of this section.

It can be readily seen that the curves of Figure 17 are not sufficiently general in application to cover all test temperatures and exposure conditions which are practical for this material. A more general strength deterioration factor has been developed for 7075-T6 which relates all three types of strength  $F_{\rm tu}$ ,  $F_{\rm ty}$ , and  $F_{\rm cy}$  (and possibly others not tested) at all test temperatures. This factor is shown plotted in Figure 20(a) against exposure value 917. In general terms, it represents a relative strength value where 1.0 is the fully hardened strength and zero is a reference (nearly annealed) strength. This strength deterioration factor, D, is:

$$D = \frac{Fef^{-F_1}}{Ff^{-F_1}} \mid Tf$$
 (5)

in which  $F_{\rm ef}$  is the strength as tested after a given exposure,  $F_{\rm f}$  is the original (fully hardened) strength,  $F_{\rm f}$  is the reference (nearly annealed) strength, and  $T_{\rm f}$  is test temperature. For a given value of D, the  $F_{\rm ef}$ ,  $F_{\rm f}$  and  $F_{\rm f}$  values must all be the same type of mechanical strength ( $F_{\rm tu}$ ,  $F_{\rm ty}$ , or  $F_{\rm cy}$ ) and obtained at the same test temperature,  $T_{\rm f}$ .

When Figure 20(a) is used as a design curve,  $F_{ef}$  is the unknown residual strength to be determined. Rearranging equation (5) for this purpose gives,

$$F_{ef} = D F_{f} + (1-D) F_{1} \mid T_{f}$$
 (6)

The appropriate value of D is obtained from Figure 20(a) for the particular total exposure value  $\theta_{17}$  (obtained as before). Appropriate values of  $F_f$  and  $F_1$  are chosen from Figure 19.

Notice that  $F_{\rm ef}$  here is for the unstressed exposure case if D is taken from Figure 20(a). The same equation applies to stressed exposure cases if appropriate values of D are substituted. These are provided in Figure 20(b) for two stressed exposure cases involving, respectively, 0.2 percent and 1.0 percent inelastic strain accumulated during exposure. The proper value of D from Figure 20(b) is used in equation (6) above with appropriate values of Ff and F1 from Figure 19 (as in the unstressed case) to find  $F_{\rm ef}$  for a stressed exposure case.

No further explicit accounting for the effects of stress during exposure is necessary. For illustration, assume a design which meets the usual aircraft design criteria of 0.2 percent inelastic strain, as substantiated by methods not covered in this report. Then, any effects on residual strength of a spectrum of conditions within the design envelope are covered by use of the D value for 0.2 percent inelastic strain. Limitations are discussed at the end of this section.

For automatic computing purposes, an analytical expression is usually preferable to an empirical curve. Referring to Figure 20(a), an expression which fits the 7075-T6 data for unstressed exposure quite well throughout most of the exposure range and which relates D to  $\theta_{17}$  is:

$$D = \frac{34,800}{(\theta_{17} - 12,920)^4 + 28,950} - 0.202 \tag{7}$$

$$(13,000 \le \theta_{17} \le 17,500)$$

This value of D can be used in place of Figure 20(a) for unstressed exposures and  $F_{\rm ef}$  can be found as before.

The added influence of stress during exposure can be taken into account by a simple correction which will suffice for many practical design situations. For example, assume that the design criteria restricts accumulated inelastic strain to 0.2 percent. From Figure 20(b) it can be estimated that for a considerable range of exposure values above  $\theta_{17} = 14,500$ , the effect of this

d cree of inelastic strain is the same as adding a small increment of exposure value, approximately

$$\Delta \theta_{17} = 100$$

The new value of D can then be found using equation (7) above with the corrected value of  $\theta_{17}$ , in this case  $\theta_{17}$  + 100, as

$$D = \frac{34,800}{(917 - 12,820)^4 + 28,950} - 0.202$$
 (8)

$$(14,500 \le 9_{17} \le 17,500)$$

If accumulated inelastic strains significantly larger than 0.2 percent are permitted, a larger correction must be made. To correct for 1.0 percent strain, the appropriate increment of  $\theta_{17}$  can be estimated from the curves of Figure 20(b) for each mechanical property. In many calculations, an average correction for the three properties will suffice and will simplify the calculations. In this case, for  $\theta_{17} \ge 14,500$ ,

$$\Delta\theta_{17}=250$$

may be used in equation (7) as in the previous case, and:

$$D = \frac{34,800}{(917 - 12,670)^4 + 28,950} - 0.202$$
 (9)

$$(14,500 \le \theta_{17} \le 17,500)$$

For  $\theta_{17}$  values between 13000 and 14,500 the appropriate curve should be chosen from Figure 20(b) to suit the design conditions. For automatic computing purposes the chosen curve can be represented either analytically as was done above or as a set of coordinate values.

The effect of inelastic strain beyond 1 percent is not considered since prior creep strains of this magnitude are not likely to be permitted (except in very local areas) in cases where short-time strength is required after creep exposure.

It should be noted that the inelastic strain values of 0.2 percent and 1.0 percent used in this report are the strain accumulated during exposure only. The yield strength tests produce an additional inelastic strain of 0.2 percent. Thus, yield strength curves for 0.2 and 1.0 percent strain during exposure actually represent 0.4 and 1.2 percent total inelastic strain, respectively. The exposure stress was considered separately to provide a consistent treatment of yield and ultimate strength results. For a vehicle designed to criteria restricting inelastic strain to 0.2 percent total from any and all sources, the above methods would be conservative.

The above methods have some limitations and also possibilities of extension. These are outlined below.

Figure 20(b) shows that there is considerable difference between the stressed exposure curves for  $F_{ty}$ ,  $F_{tu}$ , and  $F_{cy}$ , for exposures below  $\theta_{17} = 14,000$ . Here there is a very noticeable Bauschinger-type effect, which has a greater influence on strength than the factor of acceleration of the aging process through straining. In this region, only two exposure conditions were studied and the effect is only approximately defined. However, this effect is usually not taken into consideration in design except when yield strength is reduced (i.e.; when direction of testing is opposite to that of the original straining). The compressive yield curve for 1.0 percent strain can be considered an approximate minimum curve for cases of this type, providing stresses during exposure are no more severe (in causing strain) than used in the  $300^{\circ}F$ -1 hour condition tested.

Material variation must be considered in applying the results of this study. The curves of Figures 17 and 20 are representative only of the material tested in this investigation and should not therefore be used directly as minimum design curves. There is significant variation in the response of various lots of the subject alloy to exposure in addition to the normal variation in room temperature properties (see pg. 17). A suggested method of reducing strength values obtained from these curves to minimum allowable strengths is by reference to minimum design curves for 7075-T6 such as provided in Reference 7 (Mil Handbook-5). Comparison between equivalent data from this report and from the minimum design curves will indicate necessary adjustments in the curves of Figures 17 and 20. A better solution would be the development of true minimum curves of D versus 9. This could be done using the methods developed in this investigation, and, for the most part, existing data on 7075-T6.

It is felt that justification exists for cautious extension of most of the results to the case of compressive creep exposures. In the compressive creep exposure check test, 0.3 percent compressive creep strain had about the same effect on compressive yield strength as a like amount of tensile creep in the same intermediate exposure range (see Table VIII). It is not certain that like agreement would be obtained for larger amounts of strain. Also, it is probable but not certain that the Bauschinger-type effects noted for exposure values below 917 = 14,000 would be of the same magnitude for straining in compression as in tension. If Figure 20(b) is used to estimate the effect of compressive creep during exposure, the compressive yield curve should be used as the tensile yield curve, and vice-versa. This is to account for the reversal of Bauschinger-type effects.

Investigation of shear and bearing strengths was beyond the scope of this present work. However independent studies (now reported) using 7075-T6 shear and bearing data from Reference 8 have shown promise for application of Figure 20 to these failure modes. Until this is verified, shear and bearing strengths for unstressed exposures can be found from appropriate ratios to tensile values obtained for the given design conditions. It should be noted that these ratios are affected by test temperature and by approximate degree of exposure. Further studies of shear and bearing strengths after stressed and unstressed exposures are planned for further extension of this work.

Care should be exercised in applying the results of the study to conditions outside the scope of the test program. High rates of straining followed by testing in the direction opposite to that of the straining is one possible area for caution. Another such area is that of multiple tensile stressed exposures followed by testing in compression. In tests of this type, compressive yield strength was depressed slightly beyond expected values, most noticeably in the ten part sequence tests. It is not certain whether further strength loss might be caused by increasing numbers of cycles or even whether the effect is real. Care is therefore recommended in extending these results to much more complex sequences followed by compressive loading. For unstressed exposures, and for tensile properties following tensile stressed exposures the results indicate that extension to more complex sequences than those tested should be safe.

The foregoing conclusions and methods are based on the results of the test program outlined in Tables I and II. The purposes of the various tests are discussed below, then details of the program and the results obtained are presented and reviewed under the following sub-headings, in conjunction with the tables and graphs at the end of this report:

- . Test Materials and Basic Properties
- . Creep Characteristics of Material
- . Tabulated Results
- . Material Response to Thermal Expousre Without Stress
- . Exposure Parameter Re-evaluation
- . Effects of Stress During Single Exposures
- . Effects of Stress During Multiple Exposures
- . Variation of Stress Effects with Degree of Exposure
- . Residual Strength After Single Exposures
- . Accuracy of Prediction of Multiple Exposure Results
- . Generalization of Results

Each test consisted of two principal parts for all exposure conditions, single or sequential:

Exposure of Septimens -- Specimens were subjected to one of the selected temperature-time, or temperature-time-stress exposure conditions or schedules of conditions.

Strength Testing After Scoosure -- Standard short-time tensile and compressive tests were performed at room temperature or at one of the selected test temperatures.

The basic tests are the single exposure tests given in Table I. These tests provide data on stressed exposures, and they also form the foundation for examination of multiple exposures. The multiple exposure test program is presented in Table II.

Exposure temperatures, times, and stress levels shown in Table I were chosen to survey the effects of stressed exposure throughout the unstable range of the material from the T-6 condition to the nearly annealed condition. They were also chosen with the possibility in mind of using the Larson-Miller exposure parameter for stressed exposure cases as was done in Part I for unstressed exposure cases. In two cases, combinations of exposure time and temperature were approximately matched in total exposure value by two other combinations; this was done to check the assumed value of the constant C in the Larson-Miller exposure parameter. Values of this parameter using C = 17 (determination of which is discussed later) are presented for each exposure condition in the first column.

Exposure stress levels shown were established through creep tests and creep correlation methods. The target inelastic strain values were chosen to cover the range of anticipated practical use in design -- from a lower value

at half of the commonly used 0.2 percent to an upper value of 1.0 percent. Tensile exposure stress was used throughout to study stressed exposure relationships, and a spot-check case of compressive exposure stress was included.

The final test temperatures after exposure were selected to cover the temperature range of practical usage of the alloy with a minimum of tests as assisted by generalizing procedures.

Table II presents the multiple exposure tests. These tests provide data on a wide variety of sequential exposure conditions. The descriptions given in the table identify the main characteristics of the sequences chosen. Increasing and decreasing temperature trends at low, high, and zero stress levels at key values of total exposure are represented, as are some special cases with zero stress in the first or last step. Sequences range from two steps to ten steps.

Exposure temperatures and times were chosen to provide the desired total exposure and also to cause a significant change in properties in each step. The upper and lower limits of time used were 100 hrs. and 3.5 hrs., respectively, and temperatures ranged from 275°F to 400°F. Sequence total exposure values were chosen to approximately match the single exposure cases which had been checked by two different combinations of temperature and time.

Sequential exposure stress levels were established to give the indicated target strain values in the total sequence and also to produce significant strain in each of the stressed exposure steps.

Test temperatures after sequential exposures duplicated those of the single exposure program and were chosen with similar objectives. However, it was found that reductions in the number of final test temperatures could be effected without compromising objectives.

#### Test Material and Basic Properties

Test material for the entire program was 7075-T6 alclad aluminum alloy. Three .063 inches thick, 48 x 144 inch sheets of this material in the as-supplied T6 temper were used from Northrop Corporation production warehouse stock. All were from the same box of sheets from the material supplier.

The chemical analysis of each of the sheets, shown below, was well within the applicable specification (QQ-A-287) limits. Analyses were performed on a direct reading spectrograph.

#### PERCENT OF ALLOYING ELEMENTS

Element	QQ-A-287 Limits	Sheet A	Sheet B	Sheet C
Copper	1.2-2.0	1.70	1.60, 1.68	1.63
Magnesium	2.1-2.9	2.76	2.66, 2.58	2.70
Zinc	5.1-6.1	5.40	5.52, 5.48	5.50
Chromium	0.18-0.40	0.22	0.22, 0.21	0.22
Iron	0.70 max	0.10	0.18. 0.19	0.15
Silicon	0.50 max	0.06	0.05. 0.06	0.07
Manganese	0.30 max	0.10	0.09. 0.09	0.08
Titanium	0.20 max	0.02	0.02, 0.03	0.04
Aluminum	Balance	Balance	Balance	Balance

The microstructure of the test material was found to be typical of 7075 in the T-6 condition. Thickness of cladding was approximately 4 percent of the sheet thickness per side. A number of room temperature tensile tests were performed on as-received (unexposed) material from each sheet. All were well within the requirements of QQ-A-287.

Room and elevated temperature tensile and compressive properties of the test material before exposure are presented in Table III. The prefixes A, B, and C in the specimen nomenclature denote sheets A, B, and C which were utilized for the single exposure, tension; the multiple exposure, tension; and the compression program tests, respectively. The room temperature tensile tests performed on sheet A represent a survey of the sheet tensile properties. A total of 36 tensile tests were performed, to determine sheet variation and to provide tests to which nearby specimens exposed and tested could be compared. The latter improved accuracy in evaluating strength reduction factors in cases where strength values were near the original unexposed strength. The survey results are also shown in Figure 11, relating test values to location in the sheet. Variation in sheet A tensile properties was about average. The mean strength level was close to an universal mean for the same alloy and gage.

The average room temperature tensile properties of sheets B and C were approximately the same as sheet A. Room temperature compressive properties were determined only for sheet C. The average compressive yield of sheet C was 8.6 percent higher than its tensile yield, which is about normal.

Elevated temperature tensile and compressive tests in Table III are for unexposed specimens that have been heated to test temperature in a standard time of 12 minutes, from start of heating to start of loading. These data are shown plotted against temperature in Figure 19 ( $F_f$  curves).

#### Creep Characteristics of Material

Some knowledge of the creep characteristics of the material used in this study was needed to establish stress levels which would give the target values of strain during stressed exposure conditions. This need was filled by data from the literature and by check tests on each sheet of material.

Further refinements were required in methods for determining stress levels for succeeding steps in multiple exposure conditions. This was necessary because it was found that the creep strength of the material tended to decrease with progressive softening due to exposure. The strain values achieved during the various stressed single and multiple exposures are recorded in the tables of results. Analysis of creep characteristics is beyond the scope of this report.

#### Tabulated Results

The results of the experimental program are listed in Tables IV through XVI in the following order:

Table No.	Exposure Condition (Stressed & Unstressed)	Final Te <b>s</b> t	Nominal Test Temperature
IV	Single Exposure	Tension	R.T.
V	Single Exposure	Tension	200°F
VI	Single Exposure	Tension	300°F
VII	Single Exposure	Tension	400°F
VIII	Single Exposure	Compression	R.T.
IX	Single Exposure	Compression	200°F
X	Single Exposure	Compression	300°F
XI	Single Exposure	Compression	400°F
XII	Sequential Exposures	Tension	R.T.
XIII	Sequential Exposures	Tension	300°F
VIV	Sequential Exposures	Tension	400°F
vx	Sequential Exposures	Compression	R.T.
IVX	Sequential Exposures	Compression	300°F

The single exposure tables describe for each specimen the nominal and actual exposure conditions, the actual temperatures of testing after exposure, and the test results obtained. Actual exposure conditions consist of actual exposure temperature (the average temperature of the entire exposure) and the actual total inelastic strain obtained. The values of Larson-Miller exposure parameter  $\theta_{17}$  are also shown.

Similarly, actual test temperatures are distinguished from nominal; it was found that at 400°F very small deviations in test temperature could be significant, also there were a few cases where "room temperature" was significantly above the 68°F to 75°F range.

In addition to the strength values actually obtained in tensile and compressive tests, values corrected to the nominal exposure and test temperatures are provided. This was done as a guide in evaluating the effect of temperature deviations and to give a better basis for comparing results. Correction consisted of adjusting test values by an increment of strength derived from final curves of the test results themselves and appropriate to the error in exposure value ( $\theta_{17}$ ) or test temperature. This is felt to provide a reasonably accurate correction because of the large amount of data available.

The strength results are also expressed in nondimensional form in the far right hand columns of these tables. The first of these,  $R_T$  is the ratio of the test value obtained to the material original unexposed room temperature strength. The second,  $R_{\boldsymbol{\xi}}$ , is the ratio of stressed exposure test results to the parallel unstressed exposure results, or a reduction factor for the effect of stress alone in the exposure environment.

The sequential exposure tables follow the same general pattern as the single exposure tables, but modifications are made to accommodate the multiple exposure steps. Nominal exposure conditions are shown; actual exposure temperatures are omitted but were used to compute the total exposure value 917 listed for each specimen. Actual strains accumulated during each exposure step are shown below the nominal conditions. Actual strains accumulated during the entire sequences are also presented. The properties after exposures and strength ratios are treated in the same manner as in the single exposure tables.

#### Material Response to Thermal Exposure Without Stress

The results of the single exposure tension tests are plotted in Figure 12 tegether with similar results from Part I to compare the response of the two lots of material to thermal exposure. The form of the Larson-Miller parameter,  $\Theta^{\dagger}$ , established for 7075-T6 in Part I is used as the abscissa. Values of this form of the parameter are not shown in the tables because another form of the parameter is used for the balance of this report. However, values of  $\Theta^{\dagger}$  can be readily computed from the equation given in Figure 12. The ordinate is non-dimensional strength as established in Part I.

The test data plotted are from Sheet A of the present study and represent the averages of all identical tests. These data are connected with solid line curves, and the data from Part I are shown by dotted line curves.

Comparing the results from the two lots of material, it can be seen that the material from Part I responded to exposure significantly earlier in time (or in 0°) than did the material in Sheet A. Similar disagreement in proportional effect of exposure has been found in comparison of results from comparable exposure and test conditions from various sources (i.e.: Reference 8). This is a factor of material variability that must be considered in applying results of a study of this type of practical situations. Methods of approximately accounting for this variability are discussed under Analytical Methods for Design.

Another difference in results from the two studies is also illustrated in Figure 12. It was originally assumed that the  $\theta^{\dagger}$  form of the Larson-Miller parameter used in Part I would also apply to the material used in the present program. This assumption was checked by the tests in which different combinations of exposure time and temperature gave the same value of  $\theta^{\dagger}$ . Test data plotted in Figure 12 at these values of  $\theta^{\dagger}$  ( $\theta^{\dagger}=17,400$  and 18,500) show that imperfect agreement was obtained with the  $\theta^{\dagger}$  form of the parameter. Considerably better agreement was obtained when the parameter was modified. This is discussed in the following sub-section.

Best Available Copy

Reexamination of the Larson-Miller exposure parameter showed that

$$\theta_{17} = T(17 + \log t)$$

provided the best fit for the data from the material in the present study (the change to a single constant is discussed below). The results using the new constant in the parameter are shown in Figure 13. The better agreement for the ultimate strength of Sheet A is immediately apparent (the 376°F-3 hr. data point should be ignored for the moment, since it is from Sheet C and was not shown in the preceeding figure). The yield strength curves show similar good agreement for both Sheet A (tension yield points) and Sheet C (compression yield points).

There is thus the indication that the best constant for use in the parameter may change with different lots of material. The degree of change in this case can best be seen by converting the earlier  $\Theta^{\dagger}$  into the single-constant form by dividing by 1.46:

$$\frac{\theta!}{1.46} = T(13.7 + \log t)$$

Thus, the change in constant is effectively from 13.7 to 17. The change to the single constant form of the Larson-Miller parameter in this report is made because this has become the more widely used form.

#### Effects of Stress During Single Exposures

The effects of stress applied during exposure can be most simply discerned by comparing results for stressed and unstressed exposures having otherwise identical conditions. This may be expressed in ratio form as:

$$R_{\epsilon} = \frac{F_{stressed exposure}}{F_{unstressed exposure}}$$

 $R_{\mathcal{C}}$  is thus a strength reduction factor for the effects of stressed exposure, and the strain subscript  ${\mathcal{C}}$  is used because the value of this factor appears to be primarily dependent upon the amount of inelastic strain accumulated during exposure.  $R_{\mathcal{C}}$  is easily found from the test data because unstressed exposure control specimens were included with each group of three specimens subjected to stressed exposures. The control specimen test results are given in Tables IV through XI just above the associated stressed exposure results.  $R_{\mathcal{C}}$  is shown in the tables and is obtained by dividing the individual stressed exposure final strength values by the average of the control specimen final strength values.

Study of the ratios in the tables shows that exposure stress (or strain) in general does have an effect on remaining strength of 7075-T6, but this effect is not large. Re ranges from about .85 to 1.05 depending on degree of thermal exposure and degree of strain produced during exposure.

Figure 14 shows the  $K_{\epsilon}$  values from the single exposure conditions plotted against inelastic strain. Each individual test result is plotted. Separate plots are made for each strength property measured and for each degree of exposure utilized. Results for all final test temperatures are plotted together. An interesting and useful observation can be made: Final test temperature after exposure has no significant effect on the reduction factor due to stress applied during exposure. This can be confirmed by careful examination of all cases plotted in Figure 14.

In Figure 14, rough scatter bands are indicated by shading to show the basic trends of the results. These scatter bands are drawn arbitrarily in view of the small net effect that precision of establishing these bands has on the final resulting strength. The shading of the bands is stopped just above 1 percent strain because that is the upper limit of the range of interest in this study. However, points with larger strain were considered in drawing the bands.

Included in these bands are key markings (small, black "x" marks) at 0.2 percent strain and at 1.0 percent strain for use in later analysis. These marks are plotted at the approximate mean value of the test data, to the nearest 0.01 value of  $R_{\leq}$ . A small degree of smoothing of the location of these marks across neighboring exposuring conditions was done (up to 0.01  $R_{\leq}$ ). The scatter of individual test points about the key points is within 3-1/2 percent for most cases.

At the lowest value of exposure, Figure 14(a), no scatter band is established because only large-strain tests were run. Comparing the key points shows that stress during exposure had no appreciable effect on the tensile ultimate strength but it raised the tensile yield strength and depressed the compressive yield strength. This is similar to the Bauschinger effect observed in specimens which have been prestrained in tension (at room temperature) and then tested either in tension or compression (Reference 10). The comparison seems logical since the exposure stress in the 300°F-1 hour case was quite high, and a relatively large part of the resulting strain was probably plastic strain on loading.

At moderate exposure values, Figures 14(c) through 14(g), this pronounced effect has disappeared and the scatter bands for all three properties are depressed, with compressive yield reduced slightly more than the tensile properties. In this area, the predominant effect of strain during exposure seems to be an acceleration of the process of everaging or annealing. However, a small degree of Bauschinger-type effect appears to persist to fairly high exposure values.

At extreme exposure values, Figure 14(h) and 14(i), the effect of stress during exposure on remaining strength is diminished and the  $R \in$  values again approach 1.0. In this exposure range, there may be a tendency for ultimate strength to be least affected and compressive yield the most affected.

#### Affects of Stress During Multiple Exposures

The effect of creep stress during the sequential exposures is presented in Figure 15. The manner of presentation is similar to that of the single exposure results discussed in the previous section, so that visual comparisons may be made at nearly equivalent exposure values ( $\theta_{17}$  values). Direct comparisons of single and multiple exposure results are presented later in the report. In comparing the single and multiple exposure curves it is valuable to note that the vertical scale of these plots is a rather expanded one, so that any real differences should be easily detected.

Examination of the curves in Figure 15 shows that of the sequence variables considered. There is little evidence of greater strength reductions than found in comparable single exposure tests. The one sequence in which a noticably different result was obtained is the D sequence shown in Figure 15(e). In the D exposure sequence, specimens were unstressed until the last fraction of exposure period, at which time an exposure stress which gave a relatively high rate of straining was applied. Reductions in final strength caused by exposure strain in D sequence tests were not as great as in tests at similar values of strain and exposure value  $\theta_{17}$ , in which the strain was more uniformly accumulated. The results from these tests indicate that the stressed exposure strength reductions can be affected to some degree by extreme non-uniformity in strain accumulation.

Slightly greater strength reduction than expected was obtained in compression tests after the stressed E sequence (10 Part) expousre, shown in Figure 15(f). It is not certain whether this effect is real since it is small in magnitude especially considering the number of test results defining it and their scatter. The results obtained raise questions as to the effect of further increases in number of steps in sequences. Additional tests would be required to definitely establish the nature of this effect.

The various types of sequential exposures serve to demonstrate that, except for unusual exposure conditions such as the D sequence, (and possibly the case of compression after very complex stressed exposures, as noted above) the results are independent of the type of sequence used. Thus, it may be assumed for a large number of practical problems that if the sequence is sufficiently well mixed to avoid radical situations, consistent results can be expected from all sequences in the range covered by this study.

#### Variation of Stress With Degree of Exposure

As noted above, the effect of stress during exposure on the subsequent residual strength varies with degree of exposure. Figure 16 shows this variation; the key points for 0.2 percent and 1.0 percent inelastic tensile strain from Figures 14 and 15 are replotted as a function of  $\theta_{17}$  to show the trends of the mean values of the scatter bands with exposure.

The curves were drawn through the single exposure key points only (x symbol) for several reasons: (1) to establish a set of single exposure  $R_{\mathcal{E}}$  values to be used in later analysis, (2) to provide a basis to which stress

effects in multiple exposure tests could be compared, and (3) to provide a basis for some smoothing of single exposure key values across adjacent exposure conditions. Key points are omitted and dashed lines are used where  $R_{\varepsilon}$  curves had to be interplated or extrapolated to obtain values at either 0.2 percent or 1.0 percent strain. Multiple exposure key points (+ symbol) are plotted for comparison with single exposure results.

The maximum effect of exposure stress and the trend of this effect with single exposures are easily discerned from this figure, since the effects of scatter are minimized. It can be seen that for 1 percent accumulated tensile strain during exposure the maximum effect on residual strength is about 8 percent for tensile ultimate, 10 percent for tensile yield and 11 percent for compressive yield. At very early exposures, greater differences between the effects in the three cases are noted, as discussed earlier. For 0.2 percent inelastic strain, the maximum effects are about 3 percent of tensile ultimate, 4 percent of tensile yield, and 2 percent of compressive yield. The differences at 0.2 percent strain are considered smaller than the accuracy of definition of the trends.

It is of interest to note the shapes of the yield curves for 1 percent strain during exposure as compared to that of the ultimate tensile curve. The compressive yield minimum is both lower than that of the tensile yield and occurs earlier. While the tensile yield minimum even occurs later than that of the tensile ultimate. Apparently a small degree of Bauschinger-type effect persists to relatively high exposure values. Possibly, much shorter exposure periods than those investigated, with rapid accumulation of strain during exposure would emphasize this effect. However, it is significant that there was no apparent difference in this effect between 10 hour and 100 hour tests at approximately the same exposure value.

In Figure 16, the multiple exposure results follow much the same pattern as the single exposure results for tensile ultimate and yield strengths. Compressive yield, on the other hand shows a possible tendency toward more strength loss in multiple exposures than in single exposures, most noticeable in the 10 step sequence, as noted in the previous discussion. Comparison of single and multiple scatter bands at equivalent exposures for the compressive yield cases indicates that the difference is slight for the variety of cases covered. The cause of this difference, if a real effect, is not identifiable from these tests. Additional studies are desirable to check the existence of this effect.

The small net error introduced in predicting multiple exposure results for sequences within the scope of this investigation is treated in later discussions of the accuracy of prediction. The D sequence is the one case in which marked difference was obtained. In Figure 16, the D sequence points for tension ultimate and tension yield after 1 percent strain fall quite high, near the 0.2 percent strain curve, at  $\theta_{17} = 15,300$ .

#### Residual Strength After Single Exposures

In the foregoing discussion, the strength reduction factor for stressed exposure has been based on the strength after unstressed exposures. Strength after unstressed exposure is reduced as a result of: (1) duration of exposure to a sufficiently high temperature and (2) temperature at the time of failure. These effects can be combined and the total strength reduction for unstressed exposure expressed as the ratio:

 $R_T = \frac{\text{Strength at Temperature After Unstressed Exposure}}{\text{R.T.}$  Strength Before Exposure

For stressed exposure the reduction factor for the effect of stress,  $R_{\epsilon}$  is included in the expression for total strength reduction, as:

RTRe = Strength at Temperature After Stressed Exposure
R.T. Strength Before Exposure

Expressed in these terms, stressed and unstressed exposure results can be plotted on the same coordinates, providing curves that are more general than obtained by plotting actual strengths.

Figure 17 shows the  $F_{tu}$ ,  $F_{ty}$  and  $F_{cy}$  strength reduction factors  $R_T$  and  $R_TR_{\in}$  plotted as a function of exposure value  $\theta_{17}$ . Trends of  $R_T$  values are indicated by the solid curves.  $R_TR_{\in}$  values for 1.0 percent strain during exposure are shown connected by dashed curves. Each  $R_T$  data point represents an individual unstressed exposure test result. These values are also listed in Tables IV through XI. Values of  $R_TR_{\in}$  were obtained by applying  $R_{\in}$  values for 1.0 percent strain from Figure 16 to the  $R_T$  curves.

Correlation of unstressed exposure results with the Larson-Miller parameter as modified for this study appears good, as was noted previously in discussion of Figure 13. The stressed exposure results indicate that although inelastic strain during exposure does affect strength, it does not impair this correlation. This can be seen by reference to Figures 14 and 16. As noted in previous discussion of these figures, in cases where different combinations of time and temperature gave approximately the same exposure value  $\theta_{17}$ , the effect of inelastic strain on final strength was nearly the same.

The general effects of inelastic strain during exposure can be seen by examination of the curves of Figure 17 for all strength properties and final test temperatures. First, it is observed that the effect of up to 1.0 percent inelastic strain on the residual strength of 7075-T6 is not large but is significant. Second, the effect is nearly the same in all cases at a given exposure value except in the lower exposure values. A third observation is that for a large part of the exposure range the stressed exposure results are offset from the unstressed exposure results by approximately a constant increment of exposure. These observations make possible a much simplified, general approach for predicting the effect of stressed exposures presented in this report.

#### Accuracy of Prediction of Multiple Exposure Results

Use of the single exposure results to predict multiple exposure values for the same total exposure has already been suggested for unstressed exposures in Part I and for stressed exposures in foregoing discussions. Comparison of the accuracy of such predictions is given in Figure 18 for the three strength properties investigated and for unstressed and stressed exposures. The actual results achieved in test after multiple exposures are plotted against values predicted from the single exposure curves for the same calculated total 917 exposure. The solid line at 45 degrees in each chart represents perfect agreement.

Unstressed exposure correlations are shown in Figure 18(a). Individual points reflect averages of all results for each sequence. Coding is provided to identify the different sequences investigated. Agreement between predicted and actual results is in general quite good. One difference can be detected between results from certain exposure sequences. Considering sequences identical except for the temperature direction of the exposure steps, decreasing temperature steps resulted in slightly higher strengths than increasing temperature steps. This can be seen by comparing the open and closed symbols having the same shape. Even with this difference, the agreement between predicted and actual results is good.

The stressed exposure correlations are shown in Figures 18(b) and 18(c) for 0.2 percent and 1.0 percent inelastic strain. In this case the test values for comparison with predicted values were derived by using the key values of  $R_{\rm C}$  (for 0.2 and 1.0 percent strain) for the stressed multiple exposure cases from Figure 15. Each key  $R_{\rm C}$  value was multiplied by the  $R_{\rm T}$  value for the sequence it represented. This removed the scatter of individual test results as was done for Figure 18(a). Also, the  $R_{\rm T}$  values for sequences identical excepting for direction of temperature steps were averaged, removing the difference noted previously for different temperature directions of unstressed exposure sequences. (No distinction in stress effects was noted for this case, see Figures 15(a) and (c)). Since accuracy of the  $R_{\rm T}$  value prediction in Figure 18(a) was good (especially with temperature direction effect averaged), Figures 18(b) and 18(c) clearly indicate any additional inaccuracy due to the effect of stress (or strain) during the various sequences.

The results for the stressed exposures are similar to those found for the unstressed exposures. The correlation between predicted and actual test results is generally very good. The maximum difference is noticed in the case of the D sequence. Another smaller difference can be detected. This is for compressive yield strength after the ten-part E sequence. Both of these results were discussed under "Effects of Stress During Multiple Exposures."

#### Generalization of Results

It is valuable to generalize the foregoing results in order to simplify and extend application to practical problems. This can be accomplished by finding ways to relate as many variables as possible to each other on a common basis. This has already been partly accomplished by the general correlation of all test results with the form of the Larson-Miller parameter used.

Closer examination of the various curves in Figure 17 shows that all material strengths do not deteriorate to the same degree with prolonged exposure. Room temperature tensile ultimate strength declines to 50 percent of its original strength while room temperature tensile yield and compressive yield strengths decline to about 20 percent. This means that any relationship which exists between the magnitudes of these various strengths must be nonlinear and is not entirely obvious from the empirical data. However, these curves have some similarities. It can be seen that in all cases deterioration of properties begins to occur at a common degree of exposure and becomes essentially complete at another common degree of exposure. Similarity can also be seen in the general shapes of the curves. This leads to the supposition that a single general shape of decay curve might be followed by all strength properties in traversing the interval between full-hard strength and full-soft strength.

Attempts were therefore made to plot all properties tested in terms of relative strength between full-hard and full-soft strength, against the exposure parameter  $\theta_{17}$ . This was not too successful, due to a rather large degree of scatter which occurred in tests at extreme exposures (see plotted data for  $600^{\circ}\text{F-8}$  hours in Figure 17(c)). Attempts to approximate the same results by using a more restricted interval have been met with success. The interval between the original strength and the strength remaining after an exposure of  $\theta_{17} = 16,380$  (450°F-10 hr.) was chosen as the reference interval. The strength values for the extremes of this interval (for the three sheets of material used in this study) are shown in Figure 19 as a function of temperature. All other strength values are related to their position in this interval by the expression:

$$D = \frac{F_{ef} - F_1}{F_f - f_1} \bigg|_{T_e}$$

in which D is called the strength deterioration factor,  $F_{\rm ef}$  is the strength as tested after a given exposure,  $F_{\rm f}$  is the original (full hardened) strength,  $F_{\rm l}$  is the strength after the reference exposure (450°F-10 hr.) and  $T_{\rm f}$  is test temperature. For a given value of D, the  $F_{\rm ef}$ ,  $F_{\rm f}$ , and  $F_{\rm l}$  values are all the same type of mechanical strength ( $F_{\rm tu}$ ,  $F_{\rm ty}$ , or  $F_{\rm cy}$ ) and obtained at the same test temperature,  $T_{\rm f}$ .

Values of D for unstressed exposures are computed in Table XIX using data from Tables IV through XVI for  $F_{\rm ef}$  values, and Figure 19 for values of  $F_{\rm l}$  and  $F_{\rm f}$ .  $F_{\rm ef}$  values used are average values from each group of specimens tested under identical conditions. The results are plotted in Figure 20(a), where all of the unstressed exposure data from the program are shown together on one curve; all three types of mechanical properties at all final test temperatures after single and multiple exposures are represented. It can be seen that the correlation throughout most of the exposure range is good while that at extreme exposure (nearly fully annealed) is poor. Fortunately, the useful portion of the curve is the region of good correlation.

The same procedure is applied to the stressed exposure cases by relating them to the unstressed exposure results. The  $F_f$  and  $F_1$  values are unstressed exposure values, while the  $F_{ef}$  values are for stressed exposures. Calculations for D are given in Table XIX for 0.2 percent accumulated inelastic strain and for 1.0 percent accumulated inelastic strain during exposure. In these pages,  $F_{ef}$  is found by applying the strength reduction factors,  $R_e$  from Figure 16 to the same unstressed exposure average values of  $F_{ef}$  used for the basic D curve.

The results for the stressed exposure cases are shown in Figure 20(b). Here, the data points are omitted for clarity of presentation. The curves for the different properties are practically the same for most of the unstable (steeply sloping) region of the curve and are shown as a single curve for 0.2 percent and another for 1.0 percent. At the low exposure values, the stressed exposure curves for the three properties separate, particularly the 1 percent strain curves. The magnitude of the effect is considerable below  $\Theta_{17} = 13,500$ . In this region the curves are primarily based on only one exposure condition (300°F-1 hr.) and it is not certain that the same curves would hold for all exposures.

All of the test data developed in this investigation have been successfully normalized by means of the procedures which have been discussed. This raises the question of extension to mechanical properties other than those tested. It should be recognized that the normalizing procedures used in this report are basically empirical, and therefore, the resulting strength deterioration curves apply specifically to those mechanical properties for which data were obtained. However, an analysis of data from reference 8 (the results of which are not shown here) has indicated that the shear and bearing properties of 7075-T6 after various exposure conditions can very likely also be normalized by the same procedures. For unstressed exposures, shear and bearing properties may even follow the same strength deterioration curve (Figure 20(a)). The hardness data obtained in this investigation on the other hand, does not plot directly on the same curve, probably due at least in part to characteristics of hardness magnitude scales.

It is believed that the basic approach used in this work on 7075-T6 can also be used in investigating many other hardening materials which deteriorate in strength during exposure to practical thermal and stress environments. Modifications to the approach will no doubt be required, especially for materials having more complex aging or annealing characteristics than 7075-T6.

#### TEST PROCEDURES

The procedures used in this program were established and checked prior to use in an attempt to attain the accuracy desired in each part of the expessive and of subsequent tests. The consistency of the results achieved tends to verify the adequacy of the procedures. The important features of these procedures are discussed under the following subheadings:

- . Specimen Identification and Preparation
- . Specimen Exposure Assemblies
- . Specimen Exposures
- . Measurement of Creep Strain
- . Specimen Processing Between Exposure and Strength Testing
- . Hardness Testing
- Strength Testing

#### Specimen Identification and Preparation

Three 0.063 gage Alclad 7075-T6 aluminum alloy sheets were used during this investigation. Specimen blanks were marked out on heavy adhesive-backed paper covering the sheet, with the axis of all test specimens transverse to the original rolling direction. These were identified by metal stamping according to the identification system shown in Figure 1. The sheet was sheared into specimen blanks and machined to the dimensions shown in Figure 2. Careful machining practices were used, and microhardness tests on machined surfaces showed that machining caused no overheating of the material.

Specimer details are shown in Figure 2. The tension specimen incorporates pin-joint type loading ends, 0.505 inch reduced test section and 2.0 inch gage length. This same specimen was used for both creep-exposure and tensile testing after exposure. Tensile creep-exposure specimens for the compression test program were essentially extra long tensile specimens with a slight machining allowance on the width so that all edges of the compression specimens could be machined to the required tolerances. The compression test specimen was 0.5 in. wide by 2.75 in. long. Three of these specimens were machined from the reduced section of each creep-exposure specimen.

#### Specimen Exposure Assemblies

Stressed exposure of tension specimens was accomplished for the most part with three specimens linked in series as shown in Figure 3. Unstressed control specimens were exposed along with the stressed specimens, mounted (by one end only) adjacent to the center stressed specimen. Small aluminum blocks were loosely assembled to cover the reduced section of the upper and lower stressed specimens. The use of these blocks improved temperature stability with no detrimental effects other than a slightly slower heat-up rate. Thermocouples were placed at each end of the reduced section of each specimen until enough temperature data were accumulated to allow reduction of the number of thermocouples to that shown in Figure 3.

Some of the one hour and ten hour stressed exposures of tensile specimens were effected in shorter length creep furnaces one specimen at a time. Thermocouples were located at each end of the reduced section of all such specimens. No temperature stabilization blocks were required.

All thermocouples were attached so as to contact the edge of specimens. This location was checked against thermocouples located in drilled holes in a dummy specimen and found to be an accurate procedure. (Temperature deviation between edge location and hole location was found to be less than 1°F as long as reasonably uniform furnace temperatures existed. For rapid heat-up conditions, the edge thermocouples responded more rapidly to increase in furnace air temperatures than did the thermocouples in holes.)

Stressed exposure of material for compression specimens was accomplished with the creep exposure specimen shown in Figure 2. Unstressed exposure specimens consisted in most cases of 0.625 in. by 3 in. blanks. These were mounted in the center of a gecond creep exposure specimen which acted as a carrier only. The carrier specimen was mounted (by one end) parallel to the stressed exposure specimen separated by about 0.25 in. In a few cases a long creep-exposure specimen was utilized for unstressed exposure compression specimens, mounted the same as the "carrier" specimen described below.

A total of four thermocouples were used to instrument the ten-inch long reduced section of the stressed exposure specimen. A single thermocouple sufficed for the zero stressed specimens, all of which were in mutual contact.

#### Specimen Exposures

Most specimen exposures were performed in the type of creep machines shown in Figure 4. Exceptions were some short duration unstressed exposures and the compression creep exposure check test which were exposed in the circulating-air oven.

#### Creep Furnace Exposures:

Specimen assemblies were installed in the furnace and heated to the desired temperature as rapidly as possible, using caution to avoid temperature overshooting of any of the specimens. Specimens were then loaded by gently adding weights to the weight pan. The time for heat-up, from start of heating to start of loading averaged one-half hour (the creep furnaces were limited in rate at which specimens could be heated with accuracy, especially when utilizing a large proportion of the furnace length).

Thermocouple temperatures were recorded continuously throughout all exposures on multi-channel recorders and checked periodically with a precision potentiometer (the latter readings also recorded). Variation in temperature of a given point throughout the exposure period was within  $\pm$  5°F from the average temperature, after the first hour of exposure. Usually, for the first hour temperature was less stable, within about  $\pm$  7.5°F (excepting the 300°F-1 hr exposures, discussed further below). Temperature variation during exposure over specimen 2 inch gage lengths was within 3.5°F. The temperature records obtained



were used to determine specimen average temperatures for the entire exposure period. These are the actual exposure temperatures recorded in Tables of Results along with nominal exposure temperatures. Actual temperatures were up to  $\pm 4^{\circ}\mathrm{F}$  from the nominal exposure temperatures.

At the conclusion of the exposure period specimens were unloaded, removed from the furnace, disassembled from all linkage, and allowed to cool on a wood surface.

Procedures for each of the exposures in sequential exposure tests were the same as above except that specimens were usually transferred to a different creep furnace for each part of the sequence. This was done because it was found that rapid heat-up could be obtained more accurately by starting with a cool furnace.

An analysis was made of effect of heat-up time for the various exposure conditions. The conclusion was that the only exposure condition in which this became important was the 300°F-1 hr condition. The check test on the effect of compressive creep exposure (375°F-3 hr) was also short enough to have been in this category, but these exposures were performed in a circulating air oven which provided more rapid heat-up. For the short exposures in the sequential exposure program heat-up times were incorporated approximately into the calculation of total exposure for the sequence (9 value) and so were negligible in the effect on the net accuracy of the sequence.

The  $300^{\circ}$ F-1 hr exposures were given special attention and, for the tests that were reported, heat-up times were held to 15 minutes, with good temperature stability achieved throughout the exposure period. This heat-up time amounts to a possible error in exposure value of plus 50 units of  $\theta_{17}$ . The appearance of the strength reduction curves (Figure 17) at this exposure indicates that this error is negligible as to effect on strength level. Similarly, this error has a negligible effect on the determination of the effects of stress during exposure. This can best be seen in Figure 16.

## Circulating-Air Oven Exposures:

A few unstressed exposures were performed in the circulating air oven used for elevated temperature strength tests, shown in Figure 6. These were relatively short exposures, from 3 to 10 hours in duration. The specimens for each exposure were bound together and a thermocouple was attached to each end of the group. The heat-up time was 15 minutes or less except for the 600°F-8 hr exposure; one half hr was required for 600°F stabilization. Thermocouple temperatures were determined periodically with a precision potenticmeter and recorded by hand.

### Compressive Creep Exposure:

The compressive creep exposure check test followed procedures identical with the elevated temperature compression tests. The same specimen was used, supported in the compression test fixture while subjected to a predetermined compressive stress. With each of these were exposed one unstressed compression and one unstressed tension specimen as controls. Compressive creep exposure specimens were undistorted after stressed exposure and no machining was required between exposure and compression test.

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## Measurement of Greep Strain

Creep strain was measured by using two sets of gage points of the type shown in Figure 5. One had about 2m separation and the other slightly less. The larger set was impressed in the reduced section of the specimens before exposure and the smaller set was impressed within the first set after exposure. The separation between the resulting points at each end of the gage length was measured under a microscope with a Filor eyepiece, at a magnification of 100 power. The sum of the two measurements was equal to the total inelastic strain in the 2 inch gage length plus the original difference in lengths of the two sets of points.

To calibrate this system, it was necessary to know precisely the differences in lengths of the two sets of gage points. To accomplish this, each time a group of specimens for exposure was marked, trial impressions were made on a dummy strip. The differences between sets of points on the test strip was measured precisely under a microscope as before. Variation in creep strain measurement was checked and found to be within approximately  $\pm 0.02\%$  strain for the single exposure measurements.

For the long creep exposure specimens, the same procedure was followed in each of the three locations from which a compression specimen was to be machined.

For specimens subjected to multiple exposures, total creep strain accumulated was determined after each exposure step. This was done with succeeding sets of gage marks laterally displaced (up to 0.035 in.) from the original set. It was found that accuracy of creep strain measurement was reduced by the lateral displacement of gage marks. Performance checks indicated that maximum variation of measured sequential exposure strain from the actual was about 0.06%.

### Specimen Processing Between Exposure and Strength Testing

#### Tensile Specimens:

For tensile specimens, steps between exposure and tensile testing consisted of: creep strain measurement, hardness testing (if applicable, see below), and holding at room temperature for a standard delay time. The first two items are discussed elsewhere under test procedures. The standard delay time is explained as follows: Up to 1 1/2 hours were required to complete tensile tests of some groups of specimens exposed together. To minimize differences between specimens as to holding time at room temperature before testing, it was decided to delay start of testing for a standard time of one hour after the end of the exposure period. This step was taken against the possibility of minor changes in strength due to secondary ageing reactions at room temperature after exposure to temperature.

Compression Specimens:

Steps between exposure and compression testing consisted of:

Creep strain measurement
Cutting compression test blanks from reduced section of creep exposure
specimens
Hardness testing (if applicable)
Refrigeration storage of compression test blanks
Machining of compression specimens from blanks

All of the above steps are explained elsewhere or are self explanatory except the fourth. Refrigeration of compression test blanks was used to avoid long and variable soaking times at room temperature caused by the intermediate machining step. Blanks were refrigerated to below 5°F within an hour after the end of exposure and held from a minimum of 12 hours to a maximum of four weeks before bringing back to room temperature for machining and compression testing. Time at room temperature for machining and compression testing varied from 3 to 8 hours. Total time at room temperature between exposure and compression testing was therefore from 4 to 9 hours for all specimens.

The effect of refrigeration was checked on unexposed control specimens for room temperature test. Two such specimens were refrigerated for about 16 hours while several specimens located near these in the sheet were not. All were compression tested at room temperature. No effect of refrigeration could be discerned in the results (Table 3).

## Hardness Testing

All specimens for room temperature strength tests were hardness tested after exposure and before strength testing. Two Rockwell hardness scales were used because of the wide range of hardness of specimens after exposure, the B scale and the H scale. Most specimens were tested on both. Location of hardness tests on tensile specimens was at each end of the reduced section. Compression test blanks were hardness-tested at each end on material subsequently removed in machining of compression specimens.

A number of hardness test results on creep-exposed tensile specimens have been omitted from the results reported. These tests were performed just outside the specimen reduced section, through crror, on material which had been subjected to a lower exposure stress than that in the reduced section. These values would not have been completely representative of the material which was subsequently strength tested and so were deleted.

# Strength Testing

Tensile Tests:

After exposure, tensile specimens were tested at one of four test temperatures, room temperature, 200°F, 300°F or 400°F, using the equipment shown in Figure 6 for the clevated temperature tensile tests and most of those at room temperature. Properties determined were ultimate tensile strength, 0.2% offset yield strength, and percent elongation. Yield strength values were obtained from autographic stress-strain curves using extensometers over a two inch gage length. Percent elongation was measured over a two inch gage length with dividers.

Specimen temperatures were measured by a thermocouple contacting the specimen edge at the midpoint between ends of the reduced section. In tests to verify this procedure, there was no difference between readings of thermocouples located in holes drilled in the specimen and those at the edge location, nor was there any significant temperature difference from top to bottom of the reduced section. Temperature was read from a precision potentiometer. Readings were recorded at the start of loading and at maximum load. The actual test temperatures given in the tables of results are the averages of these two values, which were never more than 3°F apart.

The maximum time for heating specimens to test temperature was 12 minutes and this was for the 400°F test temperature. Less heat-up time was required for the 200°F and 300°F test temperatures, but a standard heating time of 12 minutes before loading was also used for these. Times from start of loading to the 0.2% offset yield load and to maximum load were obtained on elevated temperature tests, using a stopwatch. Time to yield varied between 20 seconds and one minute. Time from start of loading to ultimate load varied between 45 seconds and 3 1/2 minutes. (At 300°F and 400°F test temperatures maximum time was 2 1/2 minutes.) An exception to this were tests on the nearly annealed material (600°F-8 hr exposures). Some of these required 5 minutes to reach maximum load.

All tension tests were performed with a constant rate of crosshead travel from start of loading to fracture. A few of the first room temperature tensile tests were performed on a Baldwin SR-4 test machine at a strain rate of 0.005 in/in/min up to yield, with the aid of strain pacing. Shortly thereafter the Instron Machine, with a circulating air furnace (Figure 6) became available, and strength testing was changed over to this machine. The strain rate up to yield for tests conducted on the Instron Machine averaged 0.01 in/in/min. Data provided in Reference 11 indicate that a change in strain rate of this magnitude should not significantly affect the results of tensile tests on 7075-T6 at room temperature.

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Strain rates up to yield were calculated from readings of time to reach the 0.2% offset yield load. These values varied from the average value quoted above, depending on the strength level of the material and the test temperature. All strain rates fell within 0.008 and 0.012 in/in/min. This degree of variation would be undesirable for tests at 400°F on this alloy; however, nearly all of the 400°F tests fell within 0.008 and 0.01 in/in/min. The few that did not were tests of material in the nearly annealed condition. Actual strain rates up to yield for tests at 400°F are given in the tables of results (see Tables III and XVII).

Strain rate in the specimen reduced section tended to increase after yielding, the usual case when crosshead travel rate is held constant throughout tensile tests. (With specimen yielding the change in stress rate causes a change in distribution of strain between the specimen reduced section and other parts of the linkage between the test machine crossheads.) Actual strain rates from yield to ultimate were not obtained, however all fell between approximately 0.01 in/in/min and 0.05 in/in/min. The latter is the maximum possible rate with the 0.1 in/min crosshead separation rate used. Also, specimens of approximately equal strengths tested at the same test temperature would have had comparable strain rates from yield to ultimate.

Compression Tests:

After single exposures compression specimens were tested at one of four test temperatures, room temperature, 200°F, 300°F or 400°F; and after multiple exposures at room temperature or 300°F. 0.2% offset compressive yield values were obtained from autographic stress-strain curves. Test facilities were the same as for tension tests.

The fixture for supporting compression specimens during tests is shown in detail in Figure 9 and in place in the test machine in Figure 10. Specimens were supported by guide plates having offset vertical grooves. Specimen alignment in the fixture and tightening of guides were by "feel". Strain was measured by a two inch averaging extensometer, using extension arms for elevated temperature tests. Accuracy of compression test procedures was checked by inspection of modulus of elasticity values from a number of compressive and tensile stress-strain curves obtained at room temperature and 300°F.

Specimen temperatures were measured by a thermocouple attached to the edge of one of the specimen guides, at the midpoint between specimen ends, about 0.125 in. from the specimen surface. The test furnace and fixtures were stabilized at the test temperature before installing compression test specimens.

Maximum time for specimens to reach any of the test temperatures was slightly under 12 minutes, so heating time was standardized at 12 minutes for all elevated temperature compression tests. This agreed with the heating time for elevated temperature tension tests. Maximum time from start of loading to the 0.2% offset compressive yield load was 1.2 minutes.

The accuracy of compression specimen heating was checked with a dummy specimen which had thermocouples located in grooves at the top, middle and bottom in addition to the thermocouple on the specimen guide. No significant temperature disagreement occurred between any of the thermocouples when procedures which simulated compression tests were followed.

Compression tests were performed with a constant rate of crosshead travel. Strain rates as determined by stopwatch readings averaged 0.009 in/in/min for all tests. Individual values obtained varied between 0.007 and 0.001 in/in/min with a few exceptions. Part of this variation is probably due to the method of measuring strain rate. All tests at 400°F fell between 0.0075 and 0.0095 in/in/min. Strain rates for 400°F tests and for the exceptions noted above are recorded with the test results (Tables III, XI and XVI).

Equipment used in the experimental portion of this investigation can be conveniently divided into two categories.

Exposure Equipment: Equipment used to provide the desired exposure conditions for specimens prior to testing consisted of:

- . Creep-rupture machines
- . Temperature recording equipment
- . Creep strain measurement equipment

<u>Test Equipment</u>: Equipment used for tensile, compressive and hardness testing of specimens after exposure consisted of:

- . Tension and compression test equipment
- . Compression test fixture
- . Load-strain recording equipment
- . Hardness tester

### Creep-Rupture Machines

Stressed and unstressed exposure of specimens was carried out on several creep rupture machines. Six of these on which most of the exposures were performed were of the type shown in Figure 4. These are Arcweld M-3 creep frames with cylindrical wire-wound resistance furnaces, 32 inches long by 2.5 inches inside diameter. Specimen loading is by weights acting on a lever. Furnaces are controlled by Wheelco 407 current proportioning, indicating controller in conjunction with a chromel-alumel thermocouple mounted on a specimen in the center of the furnace.

Three Arcweld XJ creep frames were also used. These had 22 inch long by 4.5 inch inside diameter wire-wound resistance furnaces. Function of the XJ units is similar to the M-3 units previously discussed. Furnace control is by Brown Electronik electropulse proportioning recording controller.

### Temperature Recording Equipment

Specimen exposure temperatures were continuously recorded on one of two Brown Electronik 12-channel recorders. These recorders have a rated accuracy of 0.25 percent of full scale temperature range. The scale used was zero to  $1200^{\circ}$ F. Specimen temperatures were periodically checked with a Leeds and Northrup model 8662 precision potentiometer. Thermocouples used throughout all tests were made of special close tolerance 24 gage chromel and alumel wires joined by flash welding. Thermocouples were within  $\pm 1^{\circ}$ F accuracy in the temperature range of usage.

#### Creep Strain Measurement Equipment

Apparatus for measurement of creep strain consisted of two sets of gage points, one of approximately 2 inch separation, the other slightly less than

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2 inches. One of these is shown in Figure 5. The points used in these gage marking devices are Rockwell Diamond Point Indenters. Actual measurement of the separation of the two sets of points at each end of the 2 inch gage length was accomplished with a Bausch & Lomb bench microscope with Filor eyepiece at a magnification of 100 or 125 power.

## Tension and Compression Test Equipment

Two test machines were utilized for tension tests. A few of the first room temperature tensile tests were performed on a Baldwin SR-4 hydraulic universal testing machine, with a 50,000 pound maximum capacity. This machine is equipped with a Baldwin strain pacer operated from the extensometer differential transformer. The remainder of the tension tests and all compression tests were performed on an Instron TTC-ML test machine, shown in Figure 6. Tests on the Instron machine were run at a constant rate of crosshead travel. A stopwatch was used to measure time from start of loading to the .2 percent offset yield load and maximum load. For elevated temperature tests, a circulating air electric-resistance oven manufactured by Missimers Inc. was employed (shown in Figures 6 and 7). This oven has a working chamber 15 inches wide by 12 inches high and 22 inches deep.

Furnace temperature was controlled by a Brown Electronik electropulse proportioning recording controller, in conjunction with a chromel-alumel thermocouple located in the furnace working zone.

# Compression Test Fixture

The compression test fixture consisted of a pair of adjustable specimen guides, a loading system and a base for the specimen. Details of the fixture are shown in Figure 9. The compression test assembly is shown in Figure 10.

In this fixture, specimen guides have vertical grooves offset from each other. This has been indicated as a generally acceptable specimen supporting arrangement in Reference 9. The specimen is loaded at the top by an unattached blade which is supported by the guides and protrudes slightly at the top of the guides, where it contacts a loading ram. The blade thickness is a slight undersize of the specimen to avoid binding in the guides. Load is transmitted through a 5/8 inch diameter ball in the ram assembly to assist in alignment of loading.

Material used in the frame, guides and loading ram is A-286 alloy, with Stellite No. 1 hard facing in highly loaded areas. The loading blade is hardened tool steel. The ball is a standard ball bearing which is replaced upon any sign of plastic deformation.

### Load-Strain Recording Equipment

A Baldwin PS-5M microformer extensometer was used to measure strain in all tensile tests. This extensometer is an averaging, separable type. For tests at room temperature, 200°F and 300°F, it was attached directly to the specimen with opposing conical points, over a 2 inch gage length. For tension tests at 400°F extension arms from a different extensometer were used with the PS-5M. These were also attached to the specimen with opposing conical points

over a 2 inch gage length, with the arms extending outside the furnace. The installation for tests up to 300°F is shown in Figure 7, and for 400°F tests in Figure 8.

During compression tests strain was measured with a Baldwin SRIE resistance wire strain gage type compressometer. This is also a strain averaging instrument, and was used over a 2 inch gage length. For tests at room temperature the compressometer was attached directly to specimen edges by spring loaded knife edges. For elevated temperature tests, a set of extension arms were used which gripped the specimen edges in a similar fashion and extended outside the furnace. The SRIE was actuated by the relative motion of the extension arms, unmagnified by any lever system. The elevated temperature compression test assembly is shown in Figure 10.

Load-strain curves were obtained on the Instron machine X-Y recorder. Load was recorded on the Y axis as measured by the test machine SR-4 load cell. Strain was recorded on the X axis either from the extensometer differential transformer signal or the compressometer signal after conversion by an SR-4 converter.

## Hardness Tester

A Rockwell hardness tester, Wilson model 4JR was used with 1/16 inch diameter and 1/8 inch diameter indenters to obtain Rockwell B and H scale hardness readings. Hardness test accuracy was periodically checked by hardness tests on calibrated test blocks.

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TABLE 1
SINGLE EXPOSURE CONDITIONS TESTED

	EXPOS	URE CONI	OITIONS		FINAL TEST TEMPERATURE			
Al7 Exposure	Temp.	Time	Stress (tensile)	Target Strain	Tension	Compression		
×10-3	oF	hrs	ksi	X	o <sub>F</sub>	o <sub>F</sub>		
12.92	300	1	O	0	R.T.,300	R.T.,300		
12.92	300	1	50	1.0	R.T.,300	R.T.,300		
13.49	250	100	O.	0	R.T.,300,400	R.T.,300,400		
13.49	250	100	35-36	0.1	R.T.,300,400	R.T.,300		
13.49	250	100	<b>►.</b> (	1.0	R.T.,300,400	R.T.,300		
14.44	300	100	J	0	R.T.,200,300,400	R.T.,300		
14.44	300	100	20	0.1	R.T., 300	R.T.		
14.44	300	100	32	0.3	R.T. 200,300,400	R.T.		
14.44	300	100	34-35	1.0	R.T.,200,300,400	R.T.,300		
14.58	<b>35</b> 0	10	0	0	R.T.,300	R.T.,200,300,400		
14.58	350	10	20	0.1	R.T.,300	R.T.,300		
14.58	350	10	29	0.3	R.T.,300	R.T.,300		
14.58	350	10	30-32	1.0	R.T.,300	R.T.,300,400		
					·	No 10,500,400		
14.89	350	24	0	0	R.T.,300			
14.89	350	24	24	0.3	R.T.,300			
14.89	350	24	27	1.0	R.T.,300			
15.31	350	80	0	0	R.T.,300	R.T.,300		
15.31	350	80	13	0.1	R.T.,300	R.T.,300		
15.31	3 <b>5</b> 0	80	19	0.3	R.T.,300	R.T.,300		
15.31	350	80	21	1.0	R.T.,300	R.T.,300		
15.48	400	10	0	0	R.T.,200,300,400	R.T.,200,300,400		
15.48	$I_{i}$ 00	10	13	0.1	R.T., 200, 300, 400	R.T.,300		
15.48	400	10	17	0.3	R.T., 200, 300, 400	R.T.,300		
15.48	400	10	19-20	1.0	R.T.,200,300,400	R.T.,300,400		
8ز.16	450	10	0	0	R.T.,200,300,400	R.T.,200,300,400		
16.38	450	10	10	C.1		R.T.		
16.38	450	10	13	1.0	R.T., 200, 300	R.T.,300		
17.28	500	10	0	0	R.T. 300	R.T.,200,300,400		
17.28	500	10	9	1.0	R.T.,300	R.T.,300		
		_			·			
18.98	600	ę	0	0	R.T.,200,300,400			
		(Co	mpressive)					
14.59	375	3	0	U	R.T.	R.T.		
14.59	37 <b>5</b>	3	28	-0.3		d.T.		

TABLE II
SEQUENTIAL EXPOSURE CONDITIONS TESTED -- STRESSED EXPOSURE<sup>(1)</sup>

	SEQUENCE			EATO	BURE CONDITIONS			FINAL TEST TEMPERATURES		
Code	Description	Total 91'	Step No.	Temp.	Time	Stress (tensile)	Target Strain	Tension	Compression	
		x10-3		oł.	hra	kai		of	ok	
A-1	Decreasing temp. trend,	•	1	350	3.5	31.0	•			
	high stress	•	2	315	20	32.5	•			
		$u_{\bullet b}$	3	275	100	35.0	1.0	H.T.,300,400	н.т.,300	
A-2	Same as A-1 sequence exe	cept lower	exposu	re stres	303.		0.25	R.T.,300,400	R.T.,300	
A*-1	Inverse of A-1.	•	1	275	100	34.5	•			
	Increasing temp. trend,	•	2	315	20	32.5				
	high stress	14.6	3	350	3.5	31.5	1.0	R.T.		
A1-2	Same as A*-1 sequence es	xcept lower	exposi	are stre	sses.		0.25	R.T.		
B_12	Two steps of B-14		1	300	160	29.5				
3-12	two ecebs or p-174	15.0	2	350	21.	21.0	0.50	R.T.,300,400		
		1,00	-	7,700	~~.	2170	0.50	R. 11, 300,400		
3-13	Three steps of B-14	•	1	300	100	29.5	•			
-	Increasing temp. trend.	•	2	35u	24	21.0	•			
		15.3	3	4.00	3.5	19.0	0.75	R.T.,300,400	R.T.,300	
31-13	Inverse of B-13		1	400	3.5	21.0	•			
	Decreasing temp. trend.	•	2	350	24.	20.0	•			
		15.3	3	300	100	19.0	0.75	R.T.,300		
1-14	Mixed temp. trond,	•	1	ж	160	29.5	•			
	high stress	•	2	350	24	21.0	•			
			3	.,00	3.5	19.0	•			
		15.4	1.	150	2/.	20.0	1.0	R.T.,300,400	R.T.,300	
3-24	Same as B-14 sequence ex	ccept lower	exposi	ire atre	នុង៥៧•		0.25	R.T.,300	R.T.	
H_13=	Modified B-13	•	1	<b>30</b> 0	100	29.5	•			
J-1)4	zero-stress last-step		2	350	24	22.0				
	2010-4010D. 1200 Doop	15.3	3	4,00	3.5	U	0.50	K.T.		
111-	Modified B-14	•	1	300	100	29.5				
1-145	zero-stress last step		2	350	24	21.0	•			
	zero-scress rast scep	•	3	1,00	3.5	19.0				
		15.4	4	350	24	υ	0.75	R.T.,300	R.T.	
)	Zaro-stress first step		1	350	72	0				
•	High stress last step	15.3	2	375	3.5	25.0	1.00	R.T.,300	R.T.	
	Camaline 10 mane mirel		1	350	10	240	_			
E-1	Complex 10 part mixed	•	2	325	40	0				
	sequence	•	3	375	3.5	22.5				
		•	4	300	90	24.0	•			
		·	5	350	10	22.0	•			
			6	325	40	19.0	•			
		•	7	375	3.5	0	•			
		•	8	300	100	21.5	•			
			9	325	40	16.0	•			
		15.4	io	375	3.5	0	0.80	R.T.,300	R.T.	
		4704		• •				<del>*</del> -		
i-2	Sam: as E-l sequence exc	-					0.25	R.T.		

<sup>(1)</sup> Note: An equivalent unstressed sequential exposure program was provided by zero stress control specimens included with each exposure in this table.

TABLE III TENSILE AND COMPRESSIVE STRENGTHS -- UNEXPOSED

							TENSIC							COMPA	23510W
		ATION	HARDNESS		PROPERT TEMPER			CORAL		) 10	PITTE	CAT IO:	PROFZ	aties at Eraturs	PROPERTIES (1)
Test Test Test Of		g.u*пог +: .	Harires	Tost.	,	Fea	olor ir. z	. ,	1.,	Test	l'o.	Specieer to.	Test I-as	-,	T <sub>cy</sub>
		-	16je - 16je	٠,	- 101	pai	*	pst	F91	o,			٠,	tel	pei
a. <u>†</u> .		At at		ťO	7,75	27 <b>,</b> 90	11.0		70,100		<b>~11</b>	C11CT(3)	91	73,000	73,300
H.T.		: A155(4) : A210		75 55		9 74 <b>500</b>			78,500		511		91	72,300	72,600
K.T.		A211		- 3				5 (8),500 5 (6), <b>9</b> 00				C1108(2) C3507	91 82		73,500 75,700
a.T.	71	A2:.0		ec	υ'', 200	) 77 <b>, 300</b>	12.0	: 67,500	77,500	H.T.		C3)Uk(2)			75,100
н.î. н.î.		015A 016A		60 60				67,500			₹11	C; JUB	82		75,600
k.T.		A330		PO	67.900	70.000	11.5	5 68,000 5 68,200	78,800	,				YALLESCA.	14.,300
H.T.	71	A350		80	67,300	77,100	10.	\$ 67,600	77,500	2					
k.T. R.T.		AL10 AL30		εO				67,500 68,200							
H.T.		AU,50		+0	67,400	17,400	15.0	67,700	77,600	5					
R.T.		A520		70	49,200	77,ecc	14.0	3 69,200	77,800	)					
R.T. H.T.		A540 A566		70 67	70.200	3 77,000 3 80.600	11.0	0 64,100 5 70,000	77,000	)					
R.T.		A620		67				69,500							
R.T.	71			57	65,500	77,600	12.5	65,300	77,500	<b>)</b>					
R.T. R.T.		A660 A710		70 <b>C</b> O	/0,100 //8,500	78,900	11.5	70,100	79,300	) 1					
R.T.		A730		éo				68,300							
R.T.		A750		60	67,200	77,100	10.5	67,500	77,300	)					
R.T.		AB10 AB30		80 80	68,900	79,700	11.0	69,200	79,900	?					
R.T.	71	A850		80	67,600	77,900	11.5	67,900	76,100	,					
R.T.		A920		75	67,700	77,400	11.5	67,700	77,400	!					
R.T. R.T.		4940 4941		75 75		78,100		67,600	78,100						
R.T.	71	4960		75	68,000	78,900	10.5	65,000	78,900	)					
R.T.		Y7050		75				67,100							
R.T. A.T.		A1060 A1110		75 <b>8</b> 0	64.800	78,300	12.5	69,100	78,500	, 					
n.T.		A1130		60	69,100	78,300	10.5	69,400	78,500						
R.T.		A11 50		80				60,700							
R.T. R.T.		A1210 -		8C 80				69,000							
R.T.	102			75				70,800							
			110			Avereg	•	68,300	78,300						
R.T. R.T.		B131 B150	115.5 80.0	74	67,600	77,100	11.0	66,600 67,600	77,100						
R.T.	102	B166 (4)		77	71,200	61,100	11.5	71,200	61,100						
R.T. R.T.		8320 8160	116.0 81.0 117.0 P7.0		66,500	76,200	11.0	66,500 67,000	76,200						
X.T.	102	81266 (4)	11110 7310	้าร้า	72,900	10,500	11.0	72,900	60,500						
	_	• • •			-	Average	•	67,000	76,400						
R.T. R.T.		C210 (5)		73 91	68,300	76,100	14.0	68,300	76,300						
R.T.		C331 (5)		<b>62</b>			13.5	64,500	76,600						
		C636		73	67,900	75,100	12.5	67,900	75,100						
						Average		68,400	76,100						
200		A355		206	64,500	69,200	12.5	64,600	69,700		223	C725-1	202	69,100	69,200
200 200		A431 A420		200 200	65,000	69,700	17.0	65,000	69,700	200	223	C725-2	202	72,500	72,600
w	11	Arav		200	0),100	Average	12.0	65,100	49,900	200 200	223 223	C725-3 C725-30	203 203	11,700	
								,			_	-,-,-		lverage '	
300	71	A130		296	97,600	\$4.900	14.0	57,400	54.200	300	223	C725-4	300	63,700	(3 <b>3</b> 00
300	71			203	56,700	57,600	20.0	56,900	57,000	500		C725-5		63,600	
100	72	A720		298	56,900	54,500	21.0	56,700	56,300	300	223	C725-6	300	62,000	12.000
300		9000		299	54.300	Average 56.400	10.5	57,000 54,200	56, 100				4	Antele (	5),100
	120	n230						E1 300	S/000						
		B420		299	54,400	56,100	18.0	74,700	~1000						
					54,400 54,600	56,100 56,300	11.5	54,600	56,300						
900	120 120	B4,60		299	54,400 54,400	56,100 56,300 Average	11.5	34,400	56,300 56,200						
100	120 120 224	B420		299 300 300	54,400 54,600 56,500 55,900	56,100 56,300 Average 58,300 57,500	16.5	54,400 56,500 56,200	56,300 56,200 58,300 57,800						
900 900	120 120 224	BL20 BL60 C7217		299 300 300	54,400 54,600 56,500 55,900	56,100 56,300 Average 58,300	16.5	54,600 54,400 56,500	56,300 56,200 58,300 57,800						
900 100 100	120 120 224, 227,	BL20 BL60 C721T C721B		299 300 300 302	54,400 54,600 56,500 55,900	56,100 56,300 Average 58,300 57,500 Average	14.5 16.5 19.0	54,400 56,500 56,200 56,400	56,300 56,300 58,300 57,800 56,100	400	221	<i>c725-</i> 7(1)	400	17.300 ±	7.200
000 100 100 100	120 120 224, 227,	BL20 BL60 C721T C721B		299 300 300 302 405 198	54,400 54,400 56,500 55,900 41,400	56,100 56,300 Average 58,300 57,500 Average 43,000 45,000	11.5 16.5 19.0 14.0	54,400 56,500 56,200 56,400 42,300 43,700	56,300 56,200 58,300 57,800 56,100 43,400 44,400	400 400	20	G725-7(1) G725-0(1)	400 400	17,300 L 16,500 L	7,300 4,500
000 100 100 100	120 120 224, 227,	8,20 8,60 C7217 C7218		299 300 300 302 405 198	54,400 54,400 56,500 55,900 41,800 44,000 42,200	56,100 56,300 Average 58,300 57,500 Average 43,000 45,000 43,300	11.5 16.5 19.0 14.0	54,400 56,500 56,200 56,400 42,500 43,700 41,700	56,300 56,200 58,300 57,800 56,100 43,400 43,400 43,400	100 100 100	22) 22) 22)	G725-7(1) G725- <b>9</b> (1) G725-9(3)	100	₩,700 \	P <sub>4</sub> 700
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000 000 000 000 000 000 000	120 120 224, 227, 71 71 71 21, 274	BL20 BLA0 C721T C721B A237 (3) A266 (3) A610 (3)		299 300 300 302 405 199 401 401	56,500 55,600 55,900 41,800 42,000 42,200 42,200 42,700 42,400	56,100 56,300 Average 58,300 57,500 Average 43,000 43,100 Average 43,600 44,600	14.0 13.5 15.5 15.5	54,400 56,500 56,200 56,400 43,700 43,700 43,000 42,600 42,600	36,300 56,300 57,800 56,100 43,400 43,400 43,400 43,400 43,400 43,400 43,500 44,500	400 400 400	20 22) 22)	0725-7(1) 0725-4(1) 0725-9(3)	100	₩,700 \	P <sub>4</sub> 700
000 000 000 000 000 000 000 000	120 120 224, 224, 224, 21, 21, 21, 224, 224, 22	B4,20 B4,60 C7217 C7218 A237 (3) A266 (3) A610 (3) B212 (3) M.25 (3)		299 300 300 302 405 198 197 401 401 401	54,400 54,600 55,900 41,600 42,200 42,200 12,700 42,400 41,400	56,100 56,300 Average 58,200 57,500 Average 43,000 43,400 64,400 64,400 Average 47,600 Average 41,400 64,400 Average 41,400 64,400 64,400 64,400	11.5 16.5 19.0 14.0 13.5 15.5 13.5 16.5 17.5	54,600 54,400 56,500 56,400 42,500 43,700 41,700 42,000 42,000 42,600 42,500	56,300 50,200 58,300 57,800 57,800 13,400 14,400 13,400 14,400 14,500 14,500 14,400	400 400	22) 22) 22)	0725-7(1) 0725-4(1) 0725-9(3)	100	₩,700 \	P <sub>4</sub> 700

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<sup>(1)</sup> Properties corrected to naminal test temperature.
(2) Speciacons refrigerated prior to test.
(3) Strain rates were 0.000 to 0.000 in/in/min for tests at 100°F; 0.01) in/in/min for apoc. CilOT.
(4) Strain rates were 0.000 to 0.000 in/oness tests taken from short corners.
(5) Tensile test rerformed on a crosp speciacon. (See Figure 2)

1/ Trajection corrected to member employer emplifiens and newinal test temperature.

TABLE IV ROOM TEMP. TENSILE STRENGTH APTER SINGLE EXPOSURES

		;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;;	;	14 3 3 4 4 4 4		0.996 996 996 996		0.982 0.982 0.982 0.983	0.557 0.953 0.959 0.959	125
	AT10.	Tield .		1.055		0.975			\$ 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	
	STRENGTH RATIC.	<b>3</b> 5	0.983		6.997 0.997		0.927	00400		
	<b>E</b>	rield	C.972		7997 0.987		0.893 0.876 0.892			
	CORRECTED FROFERTIES(1)	184 12 184	7,100	77,000	77,700	7,55 200,57 200,00 1,00 1,00	72,56 72,100 73,000	44444 88888	62888 88888	33 33
	CORRE	Fty ps1	67,70	28.85 30.88	99,100	66,80 67,400 67,400 67,400	883 888	%%a38	88888 88888	2,3 2,5 3,5 3,5
		elong in 2in.	ระง ระง	85.5 8.5 8.5	11.00 20.00	55. 15. 15.	10.5		000	
	32	, t <sub>u</sub>	7,700	7,500	2,5 30,5	44.4 300 300 300 300 300 300 300 300 300 30	225 388	4444 888 888 888 888 888 888 888 888 88	62888 88888	<b>33</b>
	ETPOSI	Pty psi	67,100 67,79	588 888	3 3 8 8 8	67,700 67,700 67,700 67,900	60,100 001,100	% # 8 4 8 8 8 8 8 8	\$8238 288 288 288 288 288 288 288 288 288	e* 88
	PROPERTIES AFTER EXPOSURE	Actual Test Temp.	<b>5</b> 5	423	£%	77%	ett	2223	:43882 :43882	k k
	PROF	a	0.0		.0.0 61.0		0.85 0.45	%.0 .0	75.0 0.27	74.5
		Rockwell Rardness Ru R	11.0		115.5 116.0		115.0 115.0	u5.0	50 120 100 100	115.0
	ស	Specials Group No.	k k	kkk	κΩ	2222	225	xxxx	7222	r r
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	OSURE C	ig is	HH	444	88	8888	888	88888	<u> </u>	88
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	EC#	Speciaen No.	727		žš	5555 5	7.72	55555	33328	18
		Total X	•	888	00	3388	•••	22222	32228	38
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<b>-</b>	MONGTAL EXPOSORS CONDITIONS	70 (17-10)(d.) 210-)	8.22 8.22	222 222	67°C	รรรร กลกล	111	1111	33333	13
		# A	44	<b>444</b>	88	9999	222	22222	998888	18
		į .	ដ្ដដ	222	88	2222	888	22222	<b>12888</b> 8	Ž

TABLE IV (Continued)
ROOM TEMP. TENSILE STRENGTH AFTER SINGLE EXPOSURES

	1 ::		######################################		100 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0		0.955 0.955 0.955 0.955 0.957
37:14:	जिंदा इं	i I	Note ( )		0.935 0.935 0.935 0.935 0.935		0.092
SOUTH STUTION	#	# ( ) # # ( ) # 3		0 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1		0.77 0.727 0.083	
, <b>i</b> :)	Y. S.	2 4 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5		#2000 #2000 #2000	- <b>, , , , , , , , , , , , , , , , , , ,</b>		
ORRECTED PROPERTIES (1.)	F <sub>LU</sub>	000°5°3°	######################################	000 000 000 000 000 000 000 000 000 00	32.13 32.13 32.13 30 30 30 30 30 30 30 30 30 30 30 30 30	28.28 82.88	8888888888
CORRECTED P-OPERTIE	F <sub>Ly</sub>	57,300 57,200 58,800	333322223 3333333333333	55,780 55,780 55,780 51,580	66,500 18,000 17,000 15,300	328	********* \$88888888
	elong in Zin.	10.0	9.0 10.0 10.0 10.0 10.0 10.0	.00.00 20.00	9.5 10.5 10.5 11.0	9.0 20.5 20.5	::34453,000 0000000000000
	Ftu	20, 6% 80, 8%	26,260 26	8888 8388 8388	\$61,18 \$61,18 \$7,58	3,7,2 888	4444444444 4444444444
DOS. KE	F.y pai	57,93 8,38 30,38	28888888888888888888888888888888888888	22,22 22,23 22,53 22,53	86,44,45 80,64,44 80,164,44 81,64	33× 888	********* \$\$\$\$\$\$\$\$\$\$\$
Properties after exposure	Actual Test Temp.	<b>\$</b> \$\$	4444444	2223	23433	<b>444</b>	****
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£	h Rockwell Hardness R <sub>H</sub> R	25.0 25.0 20.0 20.0		eeee Kkkk		20.5° Rec.	ш.0
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	į r	×××	*****	8333	KRRRS	222	REFERENCE

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ed) ER SING		S ATE	Actual Test Tamp.	222	: 2221	:cex6	ಂ ಪತಕಕನ	133286	33	233	ಪ ಪ ಪ
TABLE IV (Continued) LE STRENGTH AFTER		PROPERTI		333			, , , , , , , , , , , , , , , , , , ,		22.0	3.0	
LE IV TRENG							i i	85.0		101.0	222
TABLE IV (Continued) ROOM TEMP. TENSILE STRENGTH AFTER SINGLE EXPOSURES		-	Specimen Group Ko.	225	2888	だれれいい	និនិនិនិន <u>ិ</u>	តិ <b>តិតិ</b> ននុន	ጸጸ	222	និនិនិ
MP. TE		POSTIE COMPITIONS	Strain Strain	000	6000	9001	. 00000		00	6000 677	000
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			See A	000	0000	35.53	00000	888888	٥o	585	000
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		- ji	₩								
		MODELL	1 h	222	2222	33423	22222	22222	22	222	•••

TABLE V 200<sup>0</sup>F TENSILE STRENGTH AFTER SPAGLE EXPOSURES

		To the state of th	付数の の の の の の の の の の の の の の			26.6°C	
STREETSTH RATIOS		•	5.929 5.920 5.920		0.955 0.955 0.955 0.961 0.961 0.952 0.952	6.949 0.953 0.957	
STATIST	Ü	C.785 0.791 0.815		2.52 2.52 2.55 5.55 5.55		987.3 967.3	0.155
	Tield.	0.855 0.650 0.850		0.52 5.53 0.530		0.35	\$2.0 \$2.0 \$2.0 \$2.0 \$2.0 \$2.0 \$2.0 \$2.0
ES (1.)	F <sub>tu</sub>	62,200	66.03 66.03	005,81 005,81 005,81 005,81	111122232 88888888	34,100 37,800 37,100 37,100 36,800	89,4 80,4 80,4 80,4 80,4 80,4 80,4 80,4 80
CORRECTED PROPERTIES (1)	"p #	8 8 8 8 8 6	% 4 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	37,260 39,760 36,400	2,5,5,4,5,5,5,5,5,5 5,6,6,5,5,5,5,5,5,5,5,5,5,5	2, 25 2, 26 2, 26	15,300 15,200 15,000
	clong is Zin.	:: ::: :::	22.22 22.23 22.53 25.53	15.0 20.0 16.0	15.50 15.50	15.0 16.5 16.0 13.5	15.0 15.5 16.0
31 OSO	.3. pag	62,000 62,100 63,000	3×3××× 83×833×	65,100 66,81 66,800 66,81 66,800	444442442 88888888	%% %% 888 888	% % % % %
PIOPEIETES APTER EXPOSURE	.º i	57,70 57,400 58,000	%%%%%% 8888868	77,000 39,800 36,100	22222222 22222222 22222222222222222222	44 <u>255</u> 88 888	25. 25. 28. 28. 28.
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İ	اء فإ	222	<b>8888</b> 88	8888	999999999	23 333 23 333	888

(1) Properties corrected to methal exposure conditions and seminal test temperature.

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SDIGIE E	PHOPERTIES AFTER EXPOSURE	Pet.	007°55	55,000 58,800 50,000	% % % %	******* ********* ********************	22,28 23,68 88,88	27,730 27,730 27,730 27,730 27,730 27,730 27,730	007,74 001,74 001,84	44444444444444444444444444444444444444	
Table vi 300 <sup>0</sup> f Tensile strength after single exposure	PROPERTIES	Actual Test Temp.	ន័ន័	ម្ពង់ដ	88	55 <b>8888</b>	ន័ឌ័ន័	8888888888	888	88888888888888888888888888888888888888	
T Streng		Specimen Group To.	នង	ននន	хx	aaaaaa	282	44422444	<b>22</b> 8	22277888	
NSILE	Sic	Total Strain \$	00	1.30	ေပ	11000110	000	99999999999999999999999999999999999999	000	00000000000000000000000000000000000000	
9 TE	ONDITI	Time S	44	-	88	888888	888	88888888	222	22222222	
900	ACTUAL EXPOSURE CONDITIONS	Tang.	££	%%% %%%	ន្តន	<b>888888</b>	ğğğ	######################################	222	<b>3888888</b> 488	
	ACTUAL	Spectmen No.	1017 1017	A1910 A1911 A1912	472C	PEPEE	45.5 45.3 45.3 45.3 45.3 45.3 45.3 45.3	######################################	1922 1922 1923	4922 4924 4937 4937 4930 4930	
	so.	John Strain	00	888	00	999999	000	988833555	000	9999999999	
	MCTIC COC	Stress ). ket	00	385 666	00	47.3 47.3 67.3 67.3 67.3 67.3 67.3 67.3 67.3 6	000	888 XXXXXXX 0000000000	000	8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	
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	YJZJOJ.	舞力			22	888888	888	888888888	222	2222222	
		<b>j</b>	ĸĸ	***	××	*****	XXX	REKREKREK	XXX	********	
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		i ′	4					្ទាក់ ស្ថាន និង ស្ថានសម្រាក់ ស្ថាន សមានសមានសមាន និងពីទីនៅពីពីនៅថ្នាំ
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32		elong. in 250.	22.5 23.5 10.5 17.5	17.5 17.5 11.0 11.0	23.5	200 200 200 200 200 200 200 200 200 200	80.5 17.5	84444444 6446444
EXPOSU	ostre	Ftu	25,300 25,300 25,300 25,300 25,300	113181 88888	*** 888	22222222 22222222222222222222222222222	32,200 35,200 35,200	252244222 3328333333
R SINGLE	क्राप्त य	F <sub>ty</sub> psi	2,38 2,38 2,38 2,38	2113×3 83888	35.58 36.58 86.88	**************************************	35,188 25,188 25,188	**************************************
TABLE YI (CONTINUED) 300 <sup>0</sup> F TENSILE STRENGTH AFTER SINGLE EXPOSURE	PROPERTIES AFTER EXPOSURE	Actual Testoles.	% 0 1 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	88888	888	85555555	888	8 <b>£</b> £888 <b>£88</b>
TABLE E STREN		Specimen Group No.	2244	222222	822	888888333	222	222333333
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	ACTUAL EXPOSURE CONDITIONS	,	***	288888 28888	222	22222222 22222222	ទុទ្ធទ	42224222
	ACTUAL	Specimen.	A760 B232 A855 B233	AC01 B233 A653 A655 A655 A655 A655	1997 1982 1983	535333333	333 383	24222323
	'n	Total Strair	0000	200 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	000	500000111	000	555555688
	Q POITIGE	Stress ) kat	0000	44466	ပစပ	2222224N 666666666	000	888777555 66622566
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	n	, a	านนน	<b>นนนนนน</b>	883	<b>575755575</b>	ដនដ	*****

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TABLE VI (Continued) 300°F TENGLE STRENGTE AFTER SDGLE EXPOSURE

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	re na		\$. 	0.275		0.219 0.22 0.23
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XKRECTED (1)	Fty	22224 22224 22224	85,55	26.70 26.80 26.70	17,800 17,900 17,700	388
	\$ 20.5 5 22.5.	2.78 2.00 2.00 2.00 2.00 2.00 2.00 2.00 2.0	; ;	555	3.55 5.55 5.55	37.5
16.55	7. red	25,50 26,50	27,900	30752 30752 3752	244 844 844 844	288 888
W.T. ED.	7. 2	44444 88453	32,135	16,700 16,700 16,500	96,71 20,71	288 288 288
PROPERTIES AFTER EXPOSURE	Actual Test Tump.	<u>88888</u>	ន្ត	*88	<b>258</b>	ğşş
	Species Group Fo.	ង្គង្គង្គ	t.	ង្គង្គ	444	***
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TICLIC	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	22222	ដ	222	222	* • •
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	4 4 4 W	00000	3.8	000	988	ردوه
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SIOILICOO DESOUZ TYTOIS	The left to	****	36.34	N N N	17.24 17.24	***
THE	i i	22222	я	ននង	999	***
	<b>i</b>	<b>अ</b> स्तासम	n'	ध्रध	র্গ্রয়	488

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TABLE VII 400°F TENSILE STRENGTH AFTER SINGLE EXPOSURES

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	Ü	3.5.5. 3.5.5.		in for.		#14.5 P. #1.5 #4 #1.61.61 #3.63.43		
Ŋ.	7.	883				933		75575 33333
ز. ات.		177	ន្ទដូន្តដូន្តន ១៨៧០១៧	428 428 888	EFEXY E88888	2,2 2,5 2,5 2,5 2,5 2,5 2,5 3,5 4,5 5,5 5,5 5,5 5,5 5,5 5,5 5,5 5,5 5	####### \$\$\$\$\$\$\$	<u> </u>
CORESCIENT PROPERTY SALES		1111 88111 88111	ត្តក្នុងក្នុង ខ្លួនក្នុងក្នុង	2000 2000 2000 2000 2000 2000 2000 200	****** *******	25.55 25.55	20888888 8888888	32888 82888
	Strain Pate	0.0092 0.0092 0.0065	\$6000 6000 6000 6000 6000 6000 6000 600	0,005 0,000 0,0082	500 500 500 500 500 500 500 500 500 500	0,009 0,009 0,008 0,008	2000000 200000000 20000000000000000000	2000 2000 2000 2000 2000 2000 2000 200
	elong in in	15.5 15.5 15.5	444 55.85 5.85 5.85 5.85	11.5 0.4.0	မွားမြန်နှင့် ဝင်ရန်ဆိုင်ငံ	งหูง ปล่า	*;**444 ••••••••	84433 45400
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PROPERTIES AFTER EXPOSURE	13 TH	11.12 88.123 88.23	aqaqaa 888888	87°66 , 36°6	888888 ******	77.7 888	####### \$\$\$\$\$\$\$	38888 8888
0	Actual Test Temp.	9% 503 503	<b>និន្</b> ជនិន្ន	\$. <b>3</b> \$	#8###B	888	# <b>#</b> 488#8	88488
	Specimen Group No.	20.02	222333	*52	<b>88648</b> \$	원 <sub>X</sub> -겉	8885588	a a a a a a a a a a
1055	Total Strain	၁၈၀	333858	uoo	9501111	ەبە	2238828	00000
CONDITIONS	å 1	333	888888	888	22222	ង្គ្រំង	สลลลลิสิ	2222
ACTUAL EXPOSURE	į 6	333	*****	និត្តន	*44*44	ផ្ទង្វ	544444	335 <u>6</u> 5
ACTURE.	Specialer. No.	844 8		FEFF	1111111	1.58 2.58 2.58	2325555	7555 7555 7555 7555 7555 7555 7555 755
	Strain S	000	333333 333888	<b>u</b> ua	222222 222222	uou	3335534 887758	00000
committees	Krass 27 Eri	ပဂမ	******* 888***	600	444454 88888	vov	3333888 8888888	<b></b>
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	<b>j</b>	888	KKKKKK	RRU	22222	भुस्य	4 <b>55544</b>	<b>५५५५</b> ५

il. Properties corrected to realish exposure conditions and needed test temperature.

AGO'F TENSILE STRENGTH AFTER SURGLE EXPOSURES

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	<b>7</b> -51		2000
3	ĸ <sup>3</sup>	pad	2,32 8,63 8,63 8,63 8,63 8,63 8,63 8,63 8,63
CORRECTED (1)	7.7	ned.	
!	Strain	tn/tn/ade	
	the State	×	5.4.5 5.5.5
A EXPOSITE		12	2,2,2 8,0,0 8,0,0
PROPERTIES AFTER EXPOSING	<b>7</b> ty	38	22.25 86.8
TOF	Actual Test Temp.		999
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CTUAL EDGGGE CONDITIONS	Serp. The Petal	*	000
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	3	8	355
ACTUAL	A section		8 E E
מ	Strade	•	000
CONTIN	T)	1	060
MOUTHER EXPOSERS CONTINUES	T(17-10	STA	8.88 2.83 2.83 2.83 2.83 2.83 2.83 2.83
MATTER	à		
	į	8	888

(1) Properties corrected to naminal exposure conditions and nominal test temperature.

EMP. O	TABLE VIII	ROOM TEMP. COMPRESSIVE STRENGTH A WITER SINGLE EXPOSIDE
		EMP. CO

TH BATTOS	a d	6.93 6.93 fre.0		\$1074 \$1076		5885.0 8885.0 11895.0
(1) STERIO	al'	2.989 C.993	44444 44444 44444	11013	0.927 0.927 0.939 0.999	
OCHERCIES (1) STEEN THE EATICS	P. Cy.	55 454 888 888 888	75. 75. 75. 75. 75. 75. 75. 75. 75. 75.	6 55 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	67,30 67,30 67,90 86,83 86,80	232223333 8888838888 8888888888
	Fcy	85. 45.4 86.45.4 86.45.4	75,500 75,200 75,500 75,500	85.55 86.55 86.55 86.55 86.55 86.55 86.55	65,38 88,38 88,58 88,58 88,58 88,58	5,55,568 5,53,368 5,5
PROPERTIES APTER EXPOSURE	Actual Test Temp.	55 <b>5</b> 55	5555¢	్ స్ట్ర్ట్స్ట్	<i>55556</i> 6	<b>******</b>
ROPERTIES	Pet I	80.0 79.5 80.5 80.5	22.22.23	288888 5.00.00 5.00.00	55555 000000	737777 7377 737 737 737 737 737 737 737
•	Bockwell Hardness Ru R	u7.0 u6.5 u6.5 u6.5	2,50 2,50 2,50 2,50 2,50 2,50 2,50 2,50	27.22 27.22 27.23 27.23		
	Specimen Group Re.	ee eee	166 172 173		150 179 179 164	£\$£333888
ACTUAL EXPOSURE CONDITIONS	Total Strain	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	0000	00000 00000 00000	ပပပပဝဝ	000000000000000000000000000000000000000
POSURE CA	a h	<b>44 444</b>	88388	88888	888888	888888888
ACTUAL ES	, II	298 298 299 301	255252	233 253 253 253 253 253 253 253 253 253	32555	88855555
	Specimen No.	9357 9357 9357 9357 9358	02327 02327 02337 02337	2222	9328 9228 9238 9261	000000 0000000000000000000000000000000
	Strain Strain	000	00000	44000	000000	4446699
SMOITICK	Stress ks:	ია <u>გაგ</u> აინ		36.0 27.1 27.1 27.1 3	00000	888444444 0000000000
ROCKLE EPOSTES CONTIONS	1(17%)ogt) x10°3	33 333 38 38	ងដងងង ដូដូដូដូដូ	12.49 13.49 13.49	444444 444444	********
Traca	a h	AA AAA	88888	88888	888888	88888888
	¥ -	22 222	និងអនិនិ	****	22222	888888888

TABLE VIII (Continued)
ROOM TEMP. COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

OC TR	56-585 Pt	II	49	Best Av	eldalla	C
اء يَا	8 88888	****	****	**********	333333	
# h	ងងងងងង	2272227	<b>៩</b> ឧ <b>៩៩</b> ೪೫	F5555558	22222	3
T(17-10gt)	zzzzz kkkk	**************	222222 444444	222722222 44444444444	13:55	for speci
Stress	00000	88888888 98888888	000000	22222244 66666664444	00000	en Color
Total Strair	00000	44,20000	000000	000000444	00000	8.0
Specimen Ro.	64127 64137 64137 62228 62228	CONTRACTOR	0314 0314 0314 0314 0315 0316 0316 0316 0316 0316 0316 0316 0316	<b>3333333</b> 33333333333333333333333333333	CONTRACTOR	Mernin rate for speciaen C3137 was 0.0063 in/in/atn
. <u>.</u>	32222 32222 322222	22222	**************************************	\$\$\$\$\$\$\$\$\$\$\$	88888	( rest x
Ti <b>ne</b> hr	22223	ววรวรรร	888888	ଅଞ୍ଚଳ ଅଟି ଅଟି ଅଟି	22222	rest were 0.007
Total Strain	<b>3600</b> 00	66.33	000000	000000000 39000000000000000000000000000	00000	- 0.01
Specimer Group No.	31 32 171 171 161	156 177 177 177 161 161	169 185 185 165 165	169 169 185 185 165 165 165	167 167 178 178 163	~
Abox Hard RH	25.50 25.50	22555444 25555444	ESSOCIATION SECTION SE			
Rocarell Hardress H	5. 5. 5. 5. 5. 5. 5. 5. 5. 5. 5. 5. 5. 5	12522555 000000000	61.0 61.0 62.0 62.0 62.0	8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	338888 2022 2022	
Actual Test Temp.	ម្រស់ប្រសព្	**************************************	75 88 75 75 75	<i><b>22288822</b></i>	<i>ጜጜጜጜጜ</i>	
(a. 14. (b. 14. (c. 14.	2, 13, 88 2, 13, 13, 13, 13, 13, 13, 13, 13, 13, 13	25 25 25 25 25 25 25 25 25 25 25 25 25 2	888888 141111	11111111111111 88888888888	000 001 00 000 001 00 001 001 001 001 00	
6. H	3 44 4 4 8 8 8 8 8 8 8 8	3,1%,8,8,8,8,8,8 5,35,58,6,38,8	244414 388388	242222443k	882.00 98.00 99.00 90.00	
.t:					22.4.4.2.0 2.4.4.2.0 2.4.4.2.0 2.4.4.2.0	
122,						

MUM 1 LMP. COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

OURTHES (L) STREETSTR RATIOS		30 36 30	ನ್ನ ನಿಕ್ಕ	136.3 136.3	0	) 3 5 5 5	0.00	0.375 0.381	361 365	- 4 C		7 	716.	0.277 0.283	C. 933	C.928		0.800		38°:3
CHAROTED ROPERTIES (L.)	> ○ 80 (a) (b)	12,100	206,01	35.00	39,100	8 8 8 8 8 8 8	3,13		28,300 28,700	28,800 29,200	39,485	30,6	30,700	20,500	20,100	29,300			56,126 56,136	%;; \$8,0
	Foy	306,01 306,01	30° 0° 0°	38	39,100	38,200 37,130	35,136	27,900	26,200 28,000	28,800 29,200	28,182	27,130	70° (07	20,500 21,000	3,40	19,300		39,000	25.55 38.85 38 38.85 38.85 38.85 38.85 38.85 38.85 38.85 38.85 38.85 38.85 38	57,800
PROPERTIES APTER ECPCSURE	Actual Test Temp. OF	75	ij	i) y)	۷۲ ( -	<b>S</b> (f)	5	<b>6</b> 15	100	ā! &	ផ្ល	ስነን i	ō	సస	25	č. <del>č</del> .		88	(88)	\$ <b>\$</b>
OPERTIES	ell ess R <sub>S</sub>	53.0 53.0	54.0	55.5	56.0	0.0	0.6	37.0	39.0	35	33.0	26.6	24.0							
æ.	Rockwell Hardness RH	2011 1010	0.01	110.5	111.0	286.5	109.0	105.5	105.5	104.5	15.5	125	3	% % % 5.5	8.8	8.8 5.0				
	Specimen Group No.	167 167	167	178 178	178	163	163	18. 17.	igi:	181	1 21	<u> </u>	7/.1	174	174	174		978	ส์ส์	និនិ
SHOLTICK	Total Strair	1.0 10.0	800	2.50 0.50	7.0	36	36.3	٥٥	ဝပ	0.12	33		8 0 0	00	1.15	; ř		00	, o o	% % P P
CSURE OO	Time hr	) 10 10	21	22	ន	25	32	25	122	5	42	223	2	ន្តន	96	32		60 T	<b>, m m</b>	mm
ACTUAL EXPOSURE CONDITIONS	Temp.	28	8	ផ្នុង	8	35	707	057 757	144	3,5	33	ζŞ;	757	ឌ្	3,5	78 78		37.	3.78 85.85	376 376
~	Specimen %.	2.21 2.21	CLZ1B	1.2.T	C1.27.B	0215T	22,59	51247	0126K	01257	31259	9127	9757B	0616T 0615K	56174	0617B		C610T	2610B	C510T
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	ADC TH	88			8	នុង ខ្ល	i i	4,5	344	<i>\$</i> <sub>1</sub>		<b>0</b> ਲੇਲੇ	25.	88	Š.	<b>\$</b> \$	2019 IGS	37	, k. k.	71

TABLE IX 200°F COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURES

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(1) Properties corrected to nominal exposure conditions and nominal test temperature.

TABLE X 300°F COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

	TVADACH	ECPUSIRE	HOLTHAL EXPOSEE CARDITION	ري د		ACTUAL ECPOSURE CONDITIONS	POSURE CO	NOITIONS		PHOPERTIES APTER EXPOSURE	ES SURE	OUFFECTED (TOTAL) THE PROPERTY OF	(1 1 2 2 2 2 2	Societies at
<u>i</u>	<b>1</b> 2	10.7-17	Stress t) kn5	Potal Strain	Specimen No.	Temp.	Time	Total Strain	Specimen Group No.	Actual Test <sub>o</sub> femp.	Fcy	E oy.	f.*	سبې
ន័ង		2.22	00	00	C3108 C6168	299	нн	ပပ	228 226	88	61,13 60,03 90,03	21,12 31,13	25.7 2.8.7	
888		22.22	8 % 8 0 0 0	888	C51oT C516M C516B	888	ппп	0.8 1.2 2.9 9.9	226 226 226	888	307,778 304,778 305,478	59,200 57,500 57,000		
827888	853888	7777	U 6 7 0 <b>0 0</b>	00,000	888888 88888 88888 88888 88888 88888 8888	222222	883888	<b>000000</b>	217 217 192 200 200	302 299 302 302 303	12.25 22.13 25.13 25.25 25.25 25.25 25.25	808888		
***	388888	הלילילי מהממממ	344777 300447	0000111	C6311 C631N C631B C535T C535K	######################################	888888	99888 4128888	22.0 22.0 22.0 21.7 21.7 21.7	85 3 3 3 5 5 6 6 6 6 6 6 6 6 6 6 6 6 6 6	36.53 36.53	39,200 50,500 60,500 59,100 57,000 57,000		2.463 1.063 1.086 0.95 0.955
88 888	88 888	11 111 22 22	0 0 0 0 0 0 0 0	888	C. 22.17 C. 22.17 C. 22.17 C. 22.18 C. 22.18	£ 6 8 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	38 888	00 11:33 12:23	218 218 218 218 218	361 388 388 388	55, FO 56, 30 49, 40 49, 400 49, 400 49, 400	50,100 56,500 49,100 47,400 47,400	6.755 0.752	0.882 0.871 0.841

(1) Properties cerrected to mendani exposure conditions and nominal test temperatures.

TABLE X (Continued)

\$00°F COMPRESSIVE STRENGTH AFTER SUGLE EXPOSURE

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11 \$ 7.5.2.	÷.	0.728 0.734 0.690 0.690		0.503 0.503 0.501	0.522 0.187 0.187 0.187 0.180
Outurotes Paupurotes (1 Stelente Haltus	Fcy	55,600 54,500 52,800 51,300 51,500	224 <b>88</b> 4488	8, 12, 12, 12, 13, 13, 13, 13, 13, 13, 13, 13, 13, 13	25,288 35,288 35,288 35,788 35,788 37,788 37,788 37,788 37,788 37,788 37,788
ļ	F <sub>Cy</sub>	55,600 54,100 52,200 50,800 55,100	23.78 23.78	35,780 37	66 50 50 50 50 50 50 50 50 50 50 50 50 50
PHOPZETIES AFTER ECPOSURE	Actual Test_Temp.	88888	%%% <b>%</b> %%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%	888888888888888888888888888888888888888	<b>88888 8888</b> 88888
	Specimen Group Fa.	888 888 888 888	88888888888888888888888888888888888888	25.5 25.5 25.5 25.5 25.5 25.5 25.5 25.5	<b>88884 448888</b> 88
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TABLE X (Continued)
3000r COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

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STREETS	¥.	0.355 0.351 0.355		00.250	
OORRECTED PROPERTIES (1)	Foy	26,400 26,100 26,100	807 77 807 77 807 77	26,000 19,300 26,300 19,500 26,300	19,200 19,900 19,500 27,200 20,100
	Fcy psi	26,300 26,100 26,100	2, 100 1, 200 1,	26,000 19,300 19,500 19,500	19,88 18,88
PRCPERTIES APTER EIPOSUME	Actual Test Temp.	301	300 299 301	88288 88288	<u> </u>
	Specimen Group	\$ \$ \$	\$ \$ \$ \$	23888	*****
ACTUAL EXPOSURE CONDITIONS	fotal Straft:	ပ၀၀	0.62 0.59	00000	00.00 25.00 11.00
POSUTE 0	Tian hr	222	222	22223	22222
ACTUAL D	j t	333	EEE	55888	82222
	Specimen No.	0.23 0.23 0.23 0.23	01227 01228 01220	66 14 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	55907 55908 55908 66137 66138
	Petal Serain	000	888	<b>0000</b> 0	888888
PACKL EPOSTE CONDITIONS	Stress Lat	000	13.0 13.0	<b>3000</b> 0	**************************************
ET/OSCE	T(17°10¢ t)	XXX 91 91 91	16. X 16. X 16. X	17. KKKK 17. KK 17. KK	17.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.
TADEL	4 4	999	222	2222	22222
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(1) Properties corrected to nominal exposure conditions and nominal test temperature.

TABLE XI 400°F COMPRESSIVE STRENGTE AFTER SINGLE EXPOSURE

	PORCE A	PORCYAL EDIOSURE CONDITIONS	MOLTIGNO	ø	-	ACTUAL ECPOSIES CONDITIORS	POSUME CO	MOTT TORS		PROPERTIE	PROPERTIES AFTER ELPOSURE	POSURE	CORRECTED (1) PROPERTIES STREETH RATICS	1 Smersi	H RATICS
1 P	ag ta	7(1,7-10-3 x10-3	Stress t) kai	Total Strain	Specimen Ro.	Temp.	75 FF	Total Strain	Specimen Group Ko.	Actual Test Jemp.	Fcy psi	Strain Rate in/in/min	Pcy ps:	Ę.	سيد
**	888	67:01 67:01	000	000	CJAT CJAN CJAN	252 251 251	888	ဝပပ	888	007 007	7,500 47,500 47,500	0.0090	25,500 27,500 28,100	0.626 C.639 O.547	
***	222	zzz kkr	000	000	מונג מונג מונג	322	222	000	<b>&amp;&amp;&amp;</b>	% 233	009°07	0.0065 0.0085	00°17 00°17	0.550 0.550 0.559	
888	222	***	444 555	888	SULK SULK SULK	333	222	1.83	288	33%	38,200 36,700 36,600	0.0075	36,48 36,738 36,738		0.925 0.884 0.683
488	222	13.44 14.21	000	000		888	555	000	20 5 50 5 20 5 50 5	888	30,000 30,100 39,100	0.0067	30,000 30,200 29,100	0.106 0.406 0.392	Č
888	222	222	499 888	888		8 <u>%</u> %	222	100.3	888	888	26.75 26.33 26.33	0.000 0.000 0.000 0.000	8,7,8 8,200 8,200 8,200		0.929 0.882 0.882
333	222	16. <b>2</b>	000	000	65347 65348 65348	द्रद्रद्र	222	000	222	335	889 122 122 123 123 123 123 123 123 123 123	0.0075 0.0078 0.0075	200 200 200 200 200 200 200 200 200 200	0.267	
888	222	27.72	000	000	cent sense	888	999	000	ลลล	888	17,100	0.00%	17,100	3355	

WADC TR 56-585 Pt II

TABLE XII ROOM TEMP. TENSILE STRENGTH AFTER SEQUENTIAL EXPOSURES

3	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1				0.910 C.935 0.908 C.935			0.997 0.995 0.981 0.982 0.973 0.981			.922 .905 .904 .905 .914 .925 .995 0.911 0.912 0.944
STRENUTH RUTICS	:: :::			2.932	00		6.922 6.928	000		0.691	· · · · · · · · · · · · · · · · · · ·
	Tield Tield			3.855			0.885			0.605 0.846 0.846	
2 2 2	Fee	F31		71,730	36,38 56,780		6,4 8,4 8,4	2,26 82,66 82,88		67,800 69,800 70,500	242232 8432332 88888
CORRECTED PROFERTIES (2)	F.G.	785	•	000,10	55,100		%% 8%	% % % % % %		888 888	4.28.42.2 88.8888
	• long. in 2in.	~  		9.5	30.0		9.5 U.0	20.00		999	0.0000 0.0000 0.00000
25:120	Ftu	786		27.70	%, % %, 700		%,4 %,6 %	50,65 50,65		86.55 86.68	884488 884488 884888
UTER EUR	F.	THE THE		61,000 7	55,100 ÷		7 000,00	888		%%% 8%%	25,000 25
ROPERTES APTER ELPOSURE	Actual Test Temp.	o.		g	88		ዩዩ	888		ዩዩዩ	23333
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	socines]]	7		115.0			116.5			nee:	
	Specimen Group Fo.			105	103 203		31	ដដដ		arr	និនិនិនិត
	Tage Name	~		0	3.50		00	686		000	1.4.1 5.00 5.00 5.00 5.00 5.00 5.00
	Total 0,7	3		14.57	8.8. 5.5.		%% 7.7.	222 222		22.5 24.5 25.5	******
	Step 4										
מיג אסוז	Step 3		275°F 100 hr 35.0	8	2.3	27597 100 hr 25.0	00	8	350 25.5 gr	900	202222
MICHAL EGOSUE COMPITIONS AND ABSULTING STRAIK *	Step 2		1887 1987 1987	8	0.30 0.30	# 0 # 2 # 0 # 2	00	ੂਰ । ਹੈ।	SE SE	000	*****
CTAL SOFOS	Step 1		35.5 kg	£	S1.13	35097 3.5 hr 21.0	00	8,	* K 603	000	******
	Person Be			7357	2.25 2.25 2.25	i.	5927 5927	255 255			
	18		ī			7			7-4		

Specimen iselectic strains during each step. These are listed below nominal exposure conditions.
 (Falues derived by subtraction of curilative strain resdings obtained at the end of each step, are only appreciate(s.)

(1) Astual expense temperatures (g LP7 free nominal) were used to empute  $\theta_{17}.$ 

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WADC	TR	56-585	Pt	II

NAME OF TOTAL Specimen motived Actual Pro For For a 2.2 in 2.2 in 2.2 in 2.2	Strain Group For Hardress Test Test Total in Sir. 12. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1.	2,500 5,300 30,400 30,400 0,410 30,400 30,400 30,400 30,400 30,400 30,400 30,400 30,400 30,400 30,400 30,400 30	14.54 6.15 136 7-5 55,100 08,000 13.5 55,200 6,300 13.5 55,200 10.15 13.0 13.5 13.0 13.5 13.0 13.5 13.5 13.5 13.5 13.5 13.5 13.5 13.5		15.07 0 118 112.0 56.0 3 46,600 45,400 10.0 40,400 10.0 10.0 10.0 118 117.0 11.0 118.0 17.0 17.0 17.0 17.0 17.0 17.0 17.0 17	15.02 0.40 118 73 43,700 58,100 15,5 43,700 58,100 15,5 13,700 57,100 15,5 13,700 57,100 15,5 13,700 57,100 15,5 13,700 57,100 15,5 13,700 57,100 15,5 13,700 15,5		15.25 0 125 112.0 52.0 72 11,700 57,100 11.5 11,700 57,100 0.503 (.7). 15.25 0 126 113.5 6.15 72 11,000 55,000 10.0 11,000 52,000 0.592 0.755	15.25 0.50 126 72 38,000 54,200 10.6 38,200 54,000 10.6 38,200 54,000 55,200 55		15-27 0 140 112-5 62-5 72 44,000 58,200 54,500 56,200 157 15-5 15-5 15-700 56,200 155-7 15-5 15-5 15-700 156,200 155-7 15-5 15-5 15-5 15-5 15-5 15-5 15-	:
EXELUTE STALT .  BESTLIFT STALT .  MET Stop: 1 Stop 2 Stop 5 Stop 4	Step 2 Step 5	275°F 315°F 350°F 100° m 20 hr 3.5 hr 28.0 28.0 25.0 ksi kai kai 2225 0% 05 05	1122 0.10 0.05 0 1122 0.10 0.05 0 0.05 0 0.10 0.05 0 0.1224 0.10 0.05 0	300°7 350°7 100 m 24 hr 29.5 21.0 kel kel	o o sum o o sum	1112 0.25 0.15 1112 0.20 0.15 111. 0.25 0.15	300°7 350°7 400°7 100 kr 24 kr 3.5 kr 29.5 21.0 19.0 ksi ksi	352 C C O	35.0 0.25 0.15 0.10 ±511 0.25 0.10 0.20 351. 0.25 0.05 0.15	1,00°P 350°P 300°P 3,5°ET 24°BT 100°NT 21,0°Z-0,0 19,0° 82,0°Z-0,0	M.22 C C C C C M.23 C C C C C C C C C C C C C C C C C C C	

						ROOM 1	TABLE XII (Continued) ROOM TEMP. TENSILE STRENGTE AFTER SEQUENTIAL EXPOSURES	T. ENSILE S	TABLE XII STRENGTE	CI (Con	(Continued)	UENTIA	EKPOS	URES						
	ğ	MONTHLE SPECIES CONTITIONS AND PRESENTE STRAINS	ER CONDIT.	07 SIG				rio .	ETIES A	PROPERTY ATTREBUSINE	NOSTRE.				XABECTED PROFERTIES (2)	(S)	អ	STEEVETH SATIOS	50:13	!
	a de la companya de l	See 1	Step 2	Step 3	Step 4	Total 017	Total Strade	Speciaen Group No.	Bockwell Nardness	!	Actual Test Teap.	F.y.	Ftu	elong. in Zir.	Fts	32	X.ela"	:	al m	÷
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TABLE XII (Continued) ROOM TEMP. TENSILE ETRENGTH AFTER SEQUENTIAL EXPOSURES

ļ			TEDOL	HCC	Lance T	COCCELL EXTERNATE CONDITIONS AND		MEMILIE STRAIN	زو						PRO	A.C.	PROPERTIES AFTER EDITORINE	POLITE	85	CONTROLL A	्र सर्भ		
100 A	4	Code East State 2 State 2 State 2	Stag 2	84 3	\$ 8	Nep 5	9 69 6	6 Step 7	7 3840 P	Seep 2	Step 10	O Totale,	និត	Specials Group No.	Rendra	La tra	<b>1</b>	2	e s	**		::	
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		200 200 200	usu	36.2 88.2	353	000	200	200	925	288 666	00 <b>0</b>	33.33 8.35 8.35	40.6	555	106.5 53.0 179.5 51.0 178.0 51.5	00 W	35.28 8.28 8.08 8.08	28.2 28.2 28.5	200 200 200 200 200 200 200 200 200 200		233 223 223		
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	333	588	000	ទទទ ទទ	9 8 8 6 9 9	900	888	000	950	546 666	000	15.36	22.0	255	110.0 %.9 110.5 57.0 110.5 59.0	<b>ತ</b> ತೆಪೆ ಉ೦೦	888	388 388			ទទួក ពល់វ		3,33
$\hat{x}$		And 3	i de	E L	882 102 102	inex Exex	et i Reiji	in a	885	ing.	12 C.C.	•											
	źŝ	g.,	80	80	£ c.	80	80	r.	8,	g.	ħ,	2;; 33	. 0	6 5 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	m: com	## 00	200	88	*** *** ***	22.5	613 00 66	77 70 87 70	
	2775 228	9,19	JOO	\$553 \$655	\$ 6 6 6 6 6 7	900	600	600	0 0 0 0 0 0	£ 6	000	332	444	11 M M	10.5 55.0 110.5 58.0	### 000	122	888			ម្ពង់ ពិព័ត		

1ABLE XIII 300<sup>o</sup>f Tensile Strength After Sequential Exposure

	DRU	PRICEAL ELPOSURE CONDITIONS AFE RESULTING STRAIP *	L EXPOSURE CONDITION .	CTA 2710				PROPER	PROPERTIES AFTER EXPOSURE	TPOSURE			COMBECTED PROPERTIES (2)		STHENGTH RATIOS	RUT:US	1
1 2	Section 19.	Stap 1	Step 2	Step 3	7 dens	Exposure 110	Total Strain	Specimen Group No.	Actual Test Temp.	P. J.	7. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1.	elong. in 2 ir.	Fty Ftu	Keld U	U:: Y	र सम्बद्ध	: 5
1		350°r 3.5 hr 31,500	31.5°F 20 hr 32,500	275°F 100 hr 34,500					•			t.					í .
	1005	8	8	æ		14.60	0	on	303	80,200 50,300	50,300	0.11	50,500 50,600 0.752		C.051		
		0.00 0.00 0.00	0.25 5.05	888		444 888	0.85 1.10	ដូចជ	ăăĕ	00°, 53°, 53°, 53°, 53°, 53°, 53°, 53°, 53	008,64 66,800 15,800	25.0 25.0 2.5	17,500 17,700 16,700 17,000 15,870 15,130		000	0.921	0.943 C.925 C.911
1		3.5 hr 27,500 ped	33.50% 26, hr 28,000	27,997 100 hr 29,000 ped					,								
	79.7	0	•	•		14.57	o	135	8	50,500 52,400	52,400	9.5	50,600 52,000 0.746 0.570	0.746 C	2,576		
	13 13 13 13 13 13 13 13 13 13 13 13 13 1	300	0.0 0.0 0.0 0.0 0.0	0.10 0.10		****	0.25 0.25 0.25	25 25 25 25 25 25 25 25 25 25 25 25 25 2	23.63	002.61	% % % % % % % % % % % % % % % % % % %	255 2000 2000	005'05 009'67 005'05 008'87 005'05 008'87		000	2.% 0.9%2 0.98:	0.970 0.900 0.905
27-		300 F 100 kg 29,500	350 F 22,000 24,000			٠											
	áã	აა	၀ပ			88 i.i.	ပဝ	ล์ธ์	88	12,100	12,100 13,200	18.0	12,100 43,200 0.607 12,600 43,200 0.614		0.550		
	444	222	0.15 0.15 0.10			323 11:33	67.0 67.0	និនិនិ	888	39,400	38,700 39,900 39,600 41,200 39,400 41,000	17.5 16.0 22.0	38,700 39,900 39,600 41,200 39,400 41,000			c.915 0.936 0.931	26.0 26.0 26.0 26.0

Speciars isolastic strains during each step. These are listed below nominal exposure conditions. (Whises derived by sestraction of cumulative strain residings obtained at the end of each step, are ealy appreciants.)

(1) Actual expenses temperatures (± 407 free mandant) were used to compute 917.

(2) Corrected to needand test temperatures.

	F TENSILE STRENGTS AFTER SEQUENTIAL EXPOSITE
TABLE XIII (Continued)	AFTER SEOU
TABLE XI	E STRENGTS
	SOUP TENSIL

j		,						PROPER	PROPERTIES AFTER EXPOSIPE	SOPO STURE		38 S	31.73			
	a de cina	26 p 1	Stap 2	Stap 3	7 deng	fetal 917 Exposure 110 -3	17 Tetal Strain	Specimen Group No.	Actual Test Temp.	Fty Ftu	elang. ir 2in.	·	Function is (2) ST.	STUANTS ATTOM	13. 14. 14. 14. 14. 14. 14. 14. 14. 14. 14	2
7		29.50 29.50 29.50	14 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4	4,000 3.5 F					5	psi psi	•	:Ed	FSI			ì
	200	800	· \$00	800		15.3; 15.3; 15.3;	000	<b>3</b> 3;	299	34,900 36,700 34,600 36,700	20.5	34,48 36,48	35, eQ	14 d		
	BILL BILL BILL BILL	£ & 3	0.00 0.00 0.00	0.34 0.34 0.34		15.32	200	<b>4</b> 33		34,700 36,600 32,200 34,000 32,100 34,000		32,260	200 300 300 300 300 300 300 300 300 300	8	3	
<b>8</b> -15		1.50 1.00 1.00 1.00	35 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	300°r 100 hr 10,000			<u> </u>	3		33,500 35,600		36. 36. 37. 38. 38. 38. 38. 38. 38. 38. 38. 38. 38	70.45 70.45 70.45			<u> </u>
	55.55 57.55 57.75	000	<b>00</b> 0	000		15.27	000	33	88	36,900 38,200 37,100 38,500	17.5	36,986	(1) 4(1) (0)	350		
	8311.5 1831.6 1831.7	ក្ខភ្លួក	0.15 0.15 0.10	90.09		15.28	0.45	इ इह		37,300 38,200 34,900 36,500 35,500 36,700		25. 25. 25. 25. 25. 25. 25. 25. 25. 25.	C.557	3 33 33	935	ۯ
1-1		300°7 1300 E 23°,500	25.42 100.44 100.44	3.5 tr	350% 19,500	}		ā		35,300 37,500		35,906	985	ر. ر ر	2.6.5 8.6.5	3 4 3 4 3 5
	33.5	00	00	00		15.35	00	145	300	900 36,100	22.0	37, 6750	į	į		
	7754 8756 7754	8 8 8 5 6 6	0.20	0.15 0.25 0.15	0.15 82.50 83.00	15.35	888	<b>\$</b> \$\$		34,650 35,800 31,800 33,300 31,600 33,300	13.5	31,800 33,300	0.515	0-473 0-469 0-6		
ą	naa"	300% 100 in 19,000	3300 3300 39,000	48	350% 24. br 14,000 pet			74.5		2,500 34,100	20.5	32,500 34,	34,100 34,100	ડં ઇં	0.910 0.937	3.5 5.92 5.92
_	93.50 93.54	_	٥٥	00	00	15.40	00	377	# 8 # 8	34,200 35,500	20.5	34,200 35,	0.510	997		
	25.25 25.25 25.25 25.25	60.07	0.10	900	ರಿಕರ ೮೮೦	15.37 15.37 15.37	22.25	222		33,500 34,400 32,900 34,400 33,700 35,200	27.12 17.5 10.03	33,900 35,300 33,600 34,800 32,900 34,600 33,700 35,200	905.20	0.463 0.467 0.967 0.967		288. 1.982. 1.997.

TABLE XIII (COMUNDO)
300°F TENSILE STRENGTH AFTER SEQUENTIAL EXPOSURE

TABLE XIII (COMINGO)
2007 TENSILE STRENGTH AFTER SEQUENTIAL EXPOSURE

				TYLONO	NOTE EDITOR	E COMP	TLORS AN	CONDITIONS AND RESILETING STRAIN .	THE STELL	•					PROF	STLES AL	PROFARTIES AFTER EXPOSURE	URE	SCRRE.	CORRECTED (2) PROPERTIES	STREETH RETTOS	85.748
<b>3</b>	in in the second		Kep 2 S	£ 4	7 deap	Step 5	9 6 7	Stap 7	Step 8	Step 9	Speciams Step 1 Step 2 Step 3 Step 4 Step 5 Step 6 Step 7 Step 8 Step 9 Step 10 Mo.	Deposure Strain	Tatal Straight	Spenden Group		Actual Fry F	e a €	10 Cors.	, s	12 8		Market State of the State of th
i ii	* K S %	200 M	325 m 3	2 2 3 2 4 2 3	2 2 2 2 3 3 4 4 4 4 4 4 4 4 4 4 4 4 4 4	25.04 12.04 13.04 14.04 15.04	325% 36 hr 19.0	3750p 3 hr 0 km	200 kg 200 kg 200 kg 200 kg	32.5 7.4 16.0 18.0	375 3 Er 0 Est											1
	25.	*	<b>8</b> 0	80	¥.o	¥0	ў°	80	80	80	80	15.39	ပပ	22.22	ତୁଟ୍ଲ	32,000	38,45	25.55	32,1	3.5°	2.500 636.3	4. N
	13.50 13.50	0.15 0	000	×999	0.15	0.10	88	000	ត្តព្ <u>ព</u> ុ	883	000	15-39 75-39 25-38	25.00 25.00	251 252 253 254	8 2 8 8 2 8	32,000 31,700 32,100	83 84 84 br>84 84 84 84 84 84 84 84 84 84 84 84 84 84 84 84 84 8	14.5 15.5 16.5	32,000 31,700 32,400	33,500		

Table xiv 400°f tensile strength after sequential exposures

	ğ	THE TAX	THE STATE	MOCKAL EXPOSITE COMPITIONS AND RESULTION STRAIN *				PROPE	PROPERTIES AFTER EXPOSURE	ECIP-OSUME.				CORRECTED PROPERTIES (2)	(5) ES (2)	ST	STRENGTH RATIOS	RATIOS	
3	Part i	1	Stap 1 Stap 2	\$ S	<b>*</b>	त्र का क्षेत्र क्षेत्र के कि	Servin	Specimen Group No.	Tat Tage	* 14 M	~	elong. in 2 in.	Strain Bate in/in/adn	Pty pac	rtu pad	Tield	n:.	Rexid	ult.
7		25.5.1 1.5.2.1 1.5.2.1	E K X I	12.50 14.50						ı		ł							
	707	క	8	ĸ		14.57	0	â	8	38,600	38,600	7.0	0.0078	38,600	38,500	0.574 0.495	557.0		
	41035 41036 41037	45°	200 222	0.30 2.35	ļ	7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.	1.05	<b>គ</b> គគ	33,8	35,36	35,600 36,100 37,100	9.5 9.5	c.0065 0.0065 0.0067	35,500 35,900 37,000	35,600 36,000 37,000			0,919 0,930 0,953	0.422 0.942 0.945
7		350 23.5 17.5 17.5	i sec	13.50 13.50 14.50															
	86.64	o	0	0		24.55	o	Ŕ	384	38,400	000 07	7.5	0.0093	38,100	39,700	0.563	0.51		
	133	9579	0.10 0.15 0.15	\$ 0.00		द्रहरू ततेते	0.25 0.25 0.25	នំនំនំ	88 £	36,98	36,400 37,500 36,600	5.5 5.1 5.1	0.0090	37,100 36,400 35,400	36,500 36,300			0.974 0.955 0.926	0.972 0.945 0.914
្ន រួ		I See	E ho																
	227	00	ပပ			88 77	00	ŘŘ	ផ្ទំផ្ទ	33,200	% 500 % 500 % 500	ц <u>ё</u>	0.0093	33,300 33,600	3,300 3,300	787°0	0.433 0.435		
		<b>788</b>	999			5 <b>8</b> 8 ਜਜਜ	0.35	RRR	ផ្ទផ្ទ	11,900 31,730 30,730	888 888 888	300	0.0096 0.0097 0.0094	31,90 31,800 8,000	32,500 32,300 32,300			0.92 0.949 0.945	0.947

Specimen inclustic strains during each step. These are listed below nominal exposure conditions.
 (Values derived by subtraction of commistive strain readings obtained at the end of each step, are only approximate.)

(1) Actual empowers temperatures ( $\pm k^{0} f$  from nominal) were used to compute  $\theta_{L} \gamma_{\tau}$ 

(2) Corrected to reminel test temperatures.

• •

i		TALL ROOM	MONTAL REPORTE CONDITIONS WESTITE STRAIK	DITIO'S LT.			ļ	PAOPE	PROPERTIES AFTER ECPOSURE	TPOSTIRE				CONTROLLED (2)	ES (2)	li)	STRENGTH WATTOS	AT105	1
Sec. 3	1	2 de 2	3 40 5	6 29 50	**	10 (1)	18.8 19.8 19.8 19.8 19.8 19.8 19.8 19.8	Spectaren Crour No.	Actual Test Temp.	Fty part	Ftu pai	elong. Stradi in 2 in Rate #	Strain Rate in/in/min	F <sub>2</sub> y ps:	7 g	18 "	<i>#</i>	10 mg 10 mg 10 mg 11 mg	u:.
7		Na wall	Kan H	13.55 B															
	11111	00000	00(00	00000		***** *****	00000	nness	2223	28888 28888	88888 88888	នុង្គង្គង្គ	7600°0 0°0000 0°0000 0°0000	88388 88388	88888 8888 8888 8888 8888 8888 8888 8888	3677	20.00 20.00		
	ASSECE	erees.	าววาวว	28888		222222 248242	000000 5555555 5555555	8883 <b>8</b> 8	999289	88888	828888 828888 828888	15.5 15.0 15.5 15.5	0.0000 0.0000 0.0000 0.0000 0.0000	******* ******* **********************	27.7.7.7.7. 27.7.7.7.5. 27.5.5.5.5.5.5.5.5.5.5.5.5.5.5.5.5.5.5.5		00000	0.936 0.936 0.932 0.932	0.00 0.03 0.03 0.03 0.03 0.03 0.03 0.03
7,		88.5 7 2 2 3	E the	8 % S 1	2,481 8,70														
	E530	00	00	co	o <b>o</b>	33 33 33 34 34 34 34 34 34 34 34 34 34 3	د ه	A.F.	88	8, 38 8, 28 8, 28	%;% %;%	2.5	6 <b>800</b> °3	28. 28. 28.	%, %, %, %,	0.376 C	c.330		
	225	Syn	333	332	980	222 232	717 200 200 200 200 200 200 200 200 200 20	នុន្តន	486	200 200 200 200 200 200 200 200 200 200	7,4,4 8,80 8,80 8,80 8,80 8,80 8,80 8,80	9.5	0.0072	8,83 8,83 8,83 8,83 8,83 8,83 8,83 8,83	27, 400 24, 300 24, 300 300, 42		400	1.023 5.946 0.957	1.050 0.931 0.950

56-5		NONCKAL Z	HONGHAL ERPOSIME CONDITI	DITIONS AND Ledit					PROPER	TES 47.1	PROPERTIES AFTER EXPOSITE	K:	CORRECTED (2)	;5   ;5   ;;	<i>E</i> 1
Sequence	Specials.	Step 1	Stap 2	Step 3	3 dest 4	Total 917 Exposure x10-3 (1)	Total Strain	Specimen Group Ro.	Rockwell Hardness Rp	ac.	Actual Test Temp.	rog.	Foy	a:	e:V
1		350 P 3.5 M 31.5 M	7.55 7.55 14.55 14.55	275.7 100 kg						 				† ; ;	l .
	222	80	80	80		88 33	00	176 176	15.5 16.0	76.5 76.5	2.5.	25, 530 530 530	55.49 56.586	( 0 ( 0 ( 0 ( ) ( )	
	csit csit csit		0.37 0.37	0.35		828 555	38.8 20.1 10.0 10.0 10.0 10.0 10.0 10.0 10.0	176 176 176	115.0 115.0 115.0	25.55 25.50 25.50	25.25 25.00	56,530 58,600 59,300	56,500 58,600 59,300		9.7 (v
1		35097 3.5 kg 27.5 kgs	31597 20 hr 22.0 ksd	2759 100 km 27.5 km1											
_		ក្តុង ១០០១១ ១០០១១	983 983	2000 2000 2000		28288 EEEEE	888 888 888	រុក្ខវឌ្ឍ	25.5 25.5 25.5 25.5 25.5 25.5	t to to to t to think to t t t aven	ឧឧឧឧ	31,42,83 81,52,50 81,52,50 81,52,50 81,52,50	27,22 20,23,23 20,23,23 20,23,23 20,23,23 20,23,23 20,23,23 20,23,23 20,20 20,23 20,23 20,23 20,23 20,23 20,23 20,23 20,23 20,23 20,23 20,	2 C C C C C	4.00
2.4		300°F 100°E 7.5 Est	3504 2: 17 21.5 lbs	400°F 3.5 hr 19.5 km											
:	5312 5312 5311 5311 5311 5311 5311	% <b>%</b> 4 33000	0000 1111 1111	0 0.19 0.25 0.25 0.25		22222 22448 2448	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	113 173 173 173	H 10000	62.0 61.5 57.0 56.0	22222	62,42 62,730 62,63 63,63	2 2 2 2 3 3 3 3 3 3 4 3 5 3 5 3 5 4 5 7 6 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7	0.500 0.575	
4		300°7 100 hr 20.5 ket	3519 24. hr 21.5 hai	2007 3.5 le 19.5 ket	3507 25.0 les										
	23333 25333	กรมีสิน กรมีสิน	00000	k	0 0 0 17 17 17	22.55 23.55 24.55	<b>444</b>	3333 <u>3</u>	111888 2.5.5.5	2.00 2.00 2.00 3.00 3.00 3.00 3.00 3.00	\$ \$ \$ \$ \$ \$ \$	22,500 39,300 38,700 38,700	42,300 42,500 39,300 38,700 80,000	0.569	0, 27 0, 23 0, 23 0, 23
	Specimes anchestic strains during	uc strains	Ouring sach	rep.	_	7	ALTHOUGH THE THE	eure conditions.			•				

Specimes analastic strains during each step. These are listed below number exposure conditions.
 (Values derived by selection of completive strain residings obtained at the end of each step, are ealy appreximate.)
 (1) Actual exposure temperatures (± 4°T free nominal) were used to compute 9<sub>17</sub>.
 (2) Corrected to nominal test temperatures.

CONTRACTS CATION

CORRECTED PROPERTIES (2)

PROPERTIES AFTER EXPOSURE

TABLE XV (Continued)
MOCH TEAP. COMPRESSVE STRENGTH AFTER SEQUENTIAL EXPOSURES

FC7

123

Actual Test Test.

Rockwell Randness PH Re

Speciamn Group No.

Sequence

104.1 617 Exposure (1) 22222 22424 350% 22. kg 24.5 kg5 Step 4 NORTHAL EXPOSITE CONSTITUTES AND MENUTING STRATE + 24 % 27 % 30 % 2 448 2 SESSES SESSES

21.5 kg. 21.5 kg. 20.00 20.00 20.00 21.5 kg. 21.5 kg. 00.00 00.13 00.13 00.10

0.065 0.972 0.056

27243 27443

33333 56888 8

22222

\*\*\*\*

: :

0.937

0.575 0.572

33,53,55 33,53,55 33,53,55

27,56 32,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 33,50 34,50

22222

120.0 120.0 120.0 120.0 120.0

RRRRR

**រាក្**រ **៦៦៦**១១

0.966 0.966 0.987

24444 88888 88888

22222

57.5 "8.0 \$6.5 \$6.0 \$7.0

**អ្នកអង្គ** 

C.581

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55553

WADC TR 56-585 Pt II

T. Ke

TABLE XV (Continued)
ROOM TEMP. COMPRESSIVE STRENGTE AFTER ERQUENTAL EXPOSURES

f: g		i		
7. E. T. 108	į.	Ì	. * <del>9</del> :	
74. 74.7.			3233 1771	33433
	D ii		2392 2392	\$4 <u>\$38</u> 8
3 E	3 5 12 15		្នាជា	seerer
MOUNTES ATTRA ELFOSTRE	7 5 2		88 65	444 664
Q.	Mocinal Partines		110.5	555 555 555 555 555 555 555 555 555 55
	Specimen Group Fo.		1523	EEERRR
	Total Strain		J000	೧ನಿಕ್ಕೆಗಳಿ ಕರ್ಕಾರಕ
	Total 9,7		1551	255248
	Step 10	57.00 F 7.00 18.00	3000	000000
	Step 9	2,5,5 2,0 kg.	3000	34445 34445 36666
A	9 to 16	282 282 28 E	3000	67.4887. 600000
4736 ST24	Kep 7	27.50 C Ref.		<b>0</b> 00000
S AND RESU	9 drug	25.55 25.55	3000	ក្រុកក្រុក្ខិត្ត ក្រុកក្រុកក្រុក
OCCAL ECOSCISE CONDITIONS AND RESULTING STRAIN *	ş danş	3309 3309 32.5 kgs	3000	
	Step	2007 2008 20.00 and	2000	155 <b>53</b> 3
WC30K	Step .	375°7 2 m 22.5 mai	300.,	333358 33358
	Pres i	E K.S.	ล็องอ	u <b>cas<b>o</b>a</b>
		1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	ძააა	343333
	Specials Ed.		1455 2002	######################################
	į	3		

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300°F COMPRESSIVE STRENGTH AFTER SEQUENTIAL EXPOSURES TABLE XVI

		CHI TYLOHOM	NONCHAL KIPOSUME COMDITION +	DITIONS AND AIN *					PROPERTIES APTER EXPOSURE		CORRECTED PROPERTIES (2)	STEDICT	STRENGTH PATTOC
Sequence Crode	Specians No.	же 1	Step 2	Step 3	7 6 %	1401 1401 1501 1501 1011	fotel Fredin	Specimen Group No.	Actual Test Temp.	Fcy ps:	Pai	er er	e e
7		350° 3.5 ir 31.5 kg	33.5 mi	275.7 100 fr 74.5 fr									
	CS278 CS278 CS264 CS264 CS264	ង ខេត្តប រដ្ឋា	80000 202	2882 0000 0000		4444 88688	00.78 00.78	88 88 88 188 88 88	88888	47,73 888 8388 8388	25,738 27,738 27,338 28,638	.723	\$ <b>6</b>
3		350° 3.5 to 27.5 tot	315% 20 hr 28.0 hrs	275° 100 tr 26.5 tra									
	0223 0224 0224 0224 0224	0000 1111	0000 1113	00000		***** *****	87.78 87.78	<b>5555</b> 5	<u> </u>	2,2,2,2,2 2,2,2,2,2 2,2,2,2,2 2,2,2,2,2	23,28 23,48 24,48 25,48 25,08	£5.	8. 8. 8. 8. 8. 8.
<b>87-4</b>	(1) #159	300°F 100 kr 29.5 km	350 <b>%</b> 22.5 km	2.5 km 19.5 km		;	ć	ş	\$			:	
	(C)	<b>88</b>	ង។	0.39			84	3 8 8 8 8 8 8 8	3888 8	**** 8888	8888 1888 1988 1988 1988 1988 1988 1988	i i	89. 808.
1		2007 100 hr 20.5 hrs	350°7 22.5 m	2.5 hr 3.5 hr 39.5 hats	350% Fr. 150 20.0 km								
	eeyaa Reega	កុស្តក 	7.7.7. 00000	ង <b>ពុធ</b> ០ <b>០០០</b> ០	<b>**</b> *** *** *** *** *** *** *** *** ***	22222 22222	<b>ಕೃಷ್ಣ</b> ೧೮೧೮ <b>೦</b>	1010101	2 <b>4 8 2 8</b>	36,28 32,68 32,68 32,98 32,98	36,38 36,38 37,58 32,48 32,48	667	. 316 . 897 . 889

Species inclastic strains during each step. These are listed below nominal exposure conditions.
 (Values derived by sectimation of cumulative strain restings obtained at the end of each step, are exly approximate.)

<sup>(1)</sup> Actual exposure temperatures (g 407 from nominal) were used to compute 617.

<sup>(2)</sup> Corrected to naminal test temperatures.
(3) Strain rates for specimens (51)% and (51)% were 0.013 and 0.015 in/in/min respectively.
(rest were 0.007 - 0.011)

• Specimers pulled with flat grips.
(1) Strair rate = 0.012 in/in/min

TABLE XVII TENSEON CONTROL TESTS FOR COMPRESSION PROGRAM

Trans.	E K	DOMEST RESERVED	4	ACTUAL EXFUSINE XPOITIONS	10.S				PRAPERTIES APTER ETFCSURE	APTER EUFCS	3E.		CORRECTED PROPERTIES	១នឹ	STRENCTE EATLOS	EATIOS
j .	a i	1(170)og t)	Specimen No.	ئ ئ	r ra	Specimen Group No.	Marchess Marchess RH	4 5 °	Actual Test Temp.	F <sub>ty</sub> psi	Ftu	elong in 2in.	Ft.y psi	ាដ្ឋ គ. ៤	F. Yield	Ultimate
BE		2.21	3377 •	8.8%		52.5	116.5 117.0	80.0	ಜ	88,300 900,89	77,300	9.0	36,300 36,300	77,300	1.067	1.Clt 1.628
**	ន្តន្ត	9E1	22357 •	25 25	88	221	117.c 117.0	81.5 82.5	50	57,566 57,100	35,900 35,000	11.0 11.0	07,500 07,100	35,55 30,55	C.987	566°0
ää	88	144	C3227 *	ថ្លុថ្ក	88	173	10.0 115.0	75.5 7.5	اداد	51,500 60,700	71,100	9.5	51,800 51,000	301.44 301.44	0.892	855°0
XX	22	** **	2157 C	ää	ខ្ពុខ្ព	ëë	18.0	7.0°C	233	\$5,500 \$7,200	35,56 36,56	10.5	55,200 57,900	32,48	0.865 0.846	. c.933 c.908
**	អ្ន	17 22	C7267 07268	ää	22	ää	1 1	; ;	ĭX	30,44	300,00	17.0	300°05	33. 33. 33. 33. 33. 33. 33. 33. 33. 33.	9.3 12.3	0.665 0.c58
eee	mmm	317 339	c620f •	£ 22.	<del>ጠጠ</del> ጠ	ដុំដុំដុំ	!!!	1 1 1	888	4,77 5,72 80 80 80 80 80 80 80 80 80 80 80 80 80	200,63 59,000 56,700	2 200 200	55,38 87,78	£7,300 £9,700 59,300	0.850	0.880 0.920 0.910
XX	88	15-31	STLT COURS	ន្តន្ត	8 %	185 185	הריי הריי	£2.5 61.5	88.5 82.5 84.2	201,01	% % %	8.03 0.03	6,13 81,3	35,100 55,300	0.591 0.58t	min J
និនិ	22	15.48	6227 •	88	ងង	178 178	EE.S	0.03	55 E	39,100	7 × × × × × × × × × × × × × × × × × × ×	5.5 9.5	39,100	25.43	0.572 0.572	C:115
88	22	17.22	C6157 •	ฐฐ	ន្ទន	174	8.8 	0.0 22	۲. i.,	19,700 19,300	35,300 36,600	2.52 2.53	19,700 15,300	39,300	0.288 0.282	0.516
<b>38</b>	ឧង	17.26	C7235 C7238	និនិ	22	ងង	1 1	1 ;	ध्रह	18,500 18,200	25,56 25,200	35.0 35.0	18,500	25.53 25.30	0.270 C.256	0.335
ន្តដ	អ្នក	17.28 17.28	072.5 072.5	ãã	22	**	1 1	1 1	83	15,200(1)	17,100	0.64 0.72	16,200	17,100	6.23 0.13 0.13	0.225 0.226

		ORCEIN	ORCIGINAL STRENGTH		STR	STRENGTH AFTER REFERENCE EXPOSURE (2)	r ure (2)	CHAN	CHANGE IN STRENGTH	ютв
		P.	r (1)		F <sub>1</sub> (1.)	1)		F -	F <sub>1</sub>	
	Sheet	Tenaile Yieli	Tensile Atimate	Comp.	Tensile Iteld	Tensile Ultimate	Comp. Tield	Tensile Kield	Tensile Ultimate	Comp.
g.		pei	psi	psi	psi	psī	psi	psį	psi	tea.
F -	E.H.	68,300 67,000 68,400	78,300 76,400 76,100	74,300	25,800	73,600	28,300	42,500	34,700	000,97
8	<b>∢</b> ∰∪	65,000	- - - -	70,700	24,500	38,000	27,400	40,500	31,900	7.300
86	∢¤o	57,000 54,400 56,400	88,300 56,200 58,100	63,100	24,400	28,400	26,300	32,500 30,000 _	29,900	36,800
8	<b>4</b> ฅ ဎ	42,000 42,700 43,100	43,600 44,200 44,400	- 47,500	20,300	20,400	21,600	22,000	23,200	25,900
(4)	•	1	1			!				

(1) Average Strength Values. Befer to Tables III through XI

(2) Reference Exposure = 450°F - 10 hr.

WADC	TR	56-585	Pt	11

Ä	Σ init	STOSUS: COURTINES		STRETCH	STRETCTH AT TEMPERATURE AFTER EXPOSURE	Ä	STREETH NA	STRIBIGTH DETERIORATION FACTOR	M FACTOR OSURE	STREETH S FOR S 0.25	STRENGTH DETERIORATION FACTOR FOR STRENGED EXPONURE 0.25 INFLASTIC STRAIN	N FACTOR SURE PAIN	STAENCTH S POR S	STAINGTH DETENDRATION FACTOR FOR STREASED EUROSCHE 1.05 INTLASTIC STEATS	N FACTO SECRE TEACH
				ĸ.	<b>F</b> ef (1)		a	4-74	3	Δ	Fer = F <sub>1</sub>	(2) (3)	6	1	3
į .	<b>1</b> 11	#17-16g t)		Tenaile Tield pei	Treatle Ultimate pei	Tield Paid	Tenaile Tield	Tensile Ultimate	Tield	Tenaile Yield	Tensile Ultimate	Ne. 26	Tersile Yeld	100 SEC.	長
88		8:21 2:23	18	67,100 55,900	6,24 80,000	23,700 61,100	0.972	0.968	0.946	1.83	1.013	C.98c 0.94.5	1.035	26.9 40.1	26.3
222	888	รรถก	<b>188</b>	2 % 1 800,43 800,00	5,52 86,53	75,73 86,68 84,73	0.3%	0.969 0.969 0.982	1.621 0.932 0.996	0.%9	0.908 0.969 0.982	1.005	0.98. 0.952 0.976	78.0	ក្តីក្នុង ០០០
<b>8888</b>	8888	1111	1888	3 % % % 8 % % % 8 % % %	4338 8356	67,100	0.00 4.00 6.00 6.00 6.00 6.00 6.00 6.00	0.815 0.771 0.812 0.831	0.843	0.765 0.791 0.795	0.753 0.712 0.760 0.781	0.81	2000 2000 2000 2000 2000 2000 2000 200	010 110 110 110 110 110 110 110 110 110	
2222	2222	****	1888	57,800	8 . 5 .	3843 885 885	0.752	0.729	0.73 0.73 0.73 0.73 0.73	0.712	0.669		3, 19, 19, 19, 19, 19, 19, 19, 19, 19, 19	2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	1 1333
88	สส	11.9	<b>18</b>	19,400	8,8 86,83	::	0.555 0.606	9.5 2.5 2.5 2.5 3.5 3.5 3.5 3.5 3.5 3.5 3.5 3.5 3.5 3	1 1 1 1	0.50 \$4.0	0.510		C-451	67.0	
22	88	4.3 4.3	<b>18</b>	35,68 35,68	** 88	4.50 37,70	8.0 9.70	0.37	0.352 0.31c	95.0	0.339	0.333	0.26	38.85 25.00	0.25
9999	2222	#### ####	1888	2222 8888	*2×# 8888	33%% 8888	0.328 0.295 0.345	0.337 0.262 0.260 0.331	0.20	6669 66846	0.253 0.253 0.207 0.308	0.257	33.55 5.55 5.55 5.55 5.55 5.55 5.55 5.5	0.2:1 0.19£ 0.169	0.122
3333	2222	rrrr rrrr	1888	naan Sees	3××× 8888	#### \$388	0000	0000	0000	99999 99999 99999		9668	6666	23888 3444	3333
8888	2222	17.22 17.22 17.22 17.22	1888	19,200	8; %;	8287 8588	0.135	9:16	6666 33113	91.0		6666 2353 84358	0.178	81.4	
<b>333</b> 3	••••	****	1888	85,23 85,24 85,38	2424 8666 8666		75.55 9999	6 6 6 6 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5		1111					
233	S Link	Merico etrough misso sfor to estume 5 three	threed 7 for For	31	IV through II. Tenelle data are from thest A tests only. to Table Mills for P., and Pr. P.,	for Fr	4 2 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	from Shoot A	tests onl	÷					

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TABLE XX STRENGTH DETERIORATION FACTOR FOR MULTIPLE EXPOSURES - UNSTRESSED

		0,1	STRENGTH AT IEMPERATURE AFTER EXPOSURE	EMPERATURE JRE		STRENGTH D	STRENGTH DETERIOHATION FACTOR FOR UNSTRESSED EXPOSURE	FACTOR SURE
		ļ	FOT	(2)		Da Fef	$\frac{\mathbf{Fef} - \mathbf{F_1}}{\mathbf{F_f} - \mathbf{F_1}} \tag{3}$	
Sequence	Total 917 110-3	Test Temp. Or	Tensile Yield Psi	Tensile Ultimate psi	Comp. Yield	Tensile Yield	Tensile Ultimate	Comp. Yield
<b>A-</b> 1	99°77 90°77 90°77	B. T. 300 400	61,000 50,500 33,600	71,700 50,600 38,600	65,600	0.828 0.800 0.843	0.809 0.742 0.784	0.810 0.750
A-2	7.50 7.50 7.50 7.50	R. 7. 300 400	59,900 50,600 38,100	71,500 52,400 39,700	66,000	C.802 O.803 O.820	0.804 0.802 0.831	0.819 0.739
A'-1	14.60	۳. د.	57,500	001,69	ı	576.0	0.734	ı
A1-2	09*71	ж. Т.	69,900	70,300	i	0.802	694.0	ı
<b>B-</b> 12	15.00 15.00 15.00	R. T. 300	46,800 42,400 33,500	60,400 43,200 34,200	1 1 1	0.495 0.550 0.608	767°0 767°0 787°0	1 1 1
<b>ē</b> −13	15.30 15.30 15.30	R. T. 300 400	28,900 28,700	56,800 36,600 29,400	43,300	0.376	0.388 0.292 -	0.326
B*-13	15.30	R. T. 300	43,900	58,200 38,200	Ι Ι	0.437	0.433. 0.350	1 1

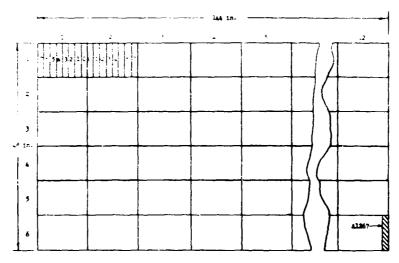
Average values of  $\Theta_s$  carried to 3 significant figures only.

Averages of individual strength values from Tables XII through XVI. Tensile specimens for A sequences and B-12 sequence were taken from sheet A. Rest of tensile specimens (above) were from Sheet B.

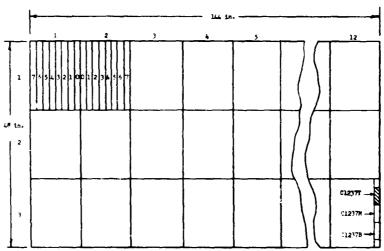
Refer to Table XVIII for values of Fl and Fr-Fl. 38 3

STRENGTH DETERIORATION FACTOR FOR MULTIPLE EXPOSURES - UNSTRESSED

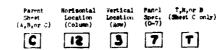
					٠			
			STREGICTH AT AFTER 1	Strength at temperature After exposure		STRENGTH DI	STRENGTH DETERIORATION FACTOR FOR UNSTRESSED EXPOSURE	FAC: <b>TO</b> R URE
			Fef	(2)	; .	E Q	Fef - Fl (	(3)
Sequence	Total \$17 110-3 (1)	Test Temp. OF	Tensile Yield psi	Tensile Ultimate psi	Comp. Yield psi	Tensile Yield	Tensile Ultimate	Comp. Yield
41 <del>-8</del>	15.40 15.40 15.40	R. T. 300 400	37,900 34,700 25,700	53,400 36,000 26,100	42,400	0.290 0.343 0.245	0.281 0.270 0.213	0.306
B-24	15.40	R. T. 300	37,900 34,100	53,000 35,500	42,800	0.290	0.269	0.315
B-13a	15.30	R. T.	001,14	56,200		0.368	0,369	ı
B-14e	15.40	ж. т. 300	39,600 33,900	54,300 35,100	75,500	0.332	0.309	905.0
Ω	15.30 15.30	R. T. 300	44,200 37,500	58,800 38,800	75,900	0.444	0.450	0.317
<b>F-1</b>	15.40	ж. т. 300	38,900	54,300 36,100	42,400	0.315	0.309	0.306
3-2	15.40	R. T.	39,900	24,600	•	0.339	0.319	•
<u>2</u> 3	15.40	R. T.	38,700	53,900	ı	0.310	252.0	ı



(a) Tension Specimen Identification, Sheets A and B



(b) Compression Specimen Identification, Sheet C



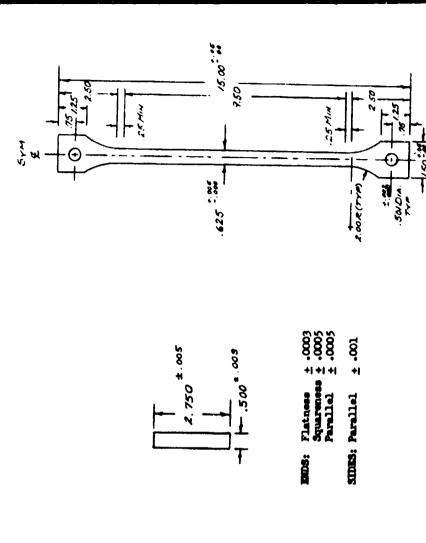
(c) Specimen Numbering Code

FIGURE 1 IDENTIFICATION OF TENSION AND COMPRESSION SPECIMENS

COMPRESSION SPECIMEN

TENSILE SPECTIEN

CREEP EXPOSURE SPECIMEN



Takery

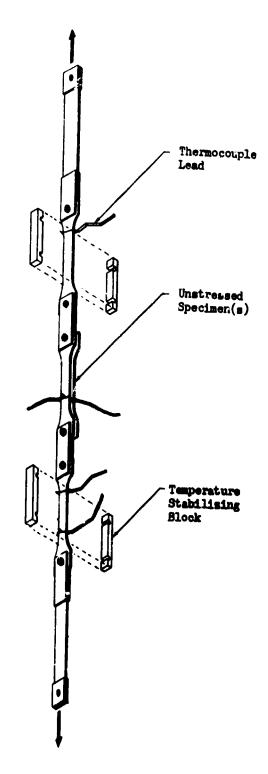
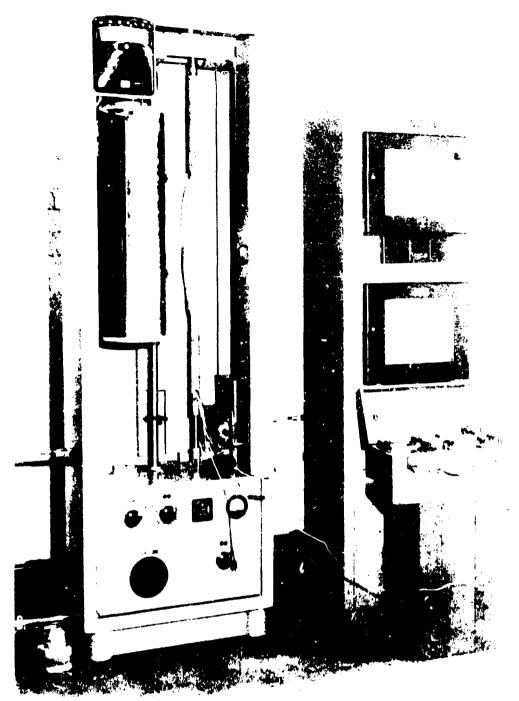
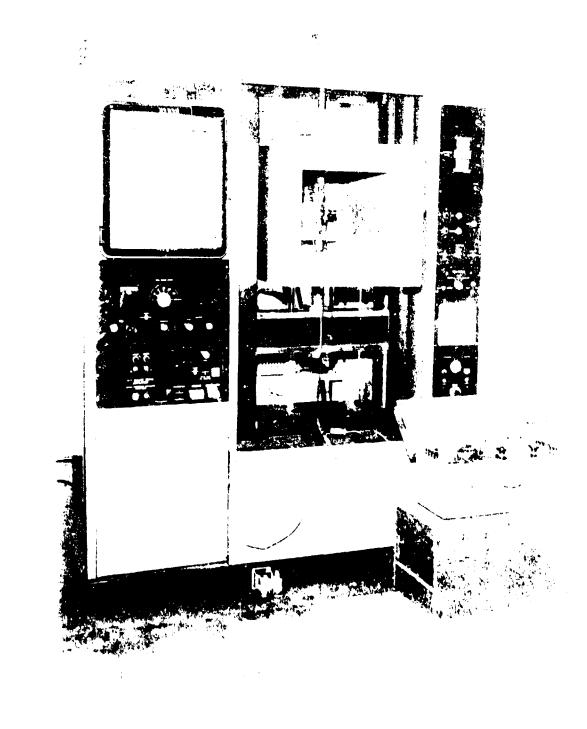


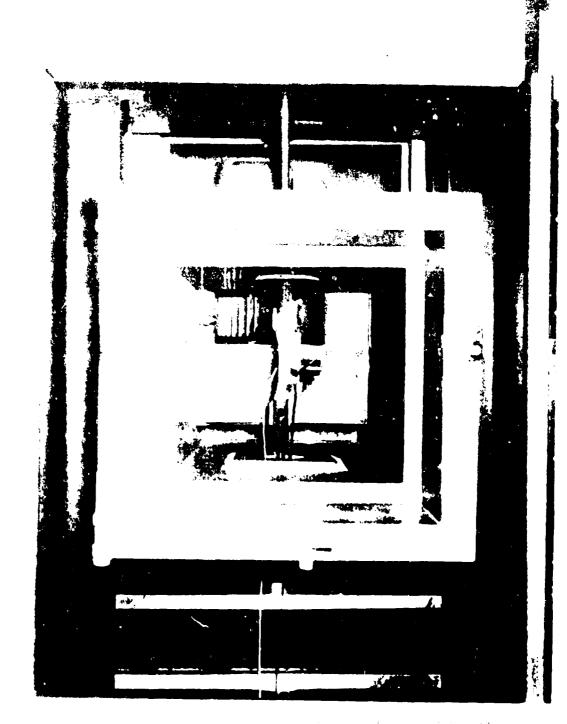
FIGURE 3 TENSION SPECIMEN EXPOSURE ASSEMBLY

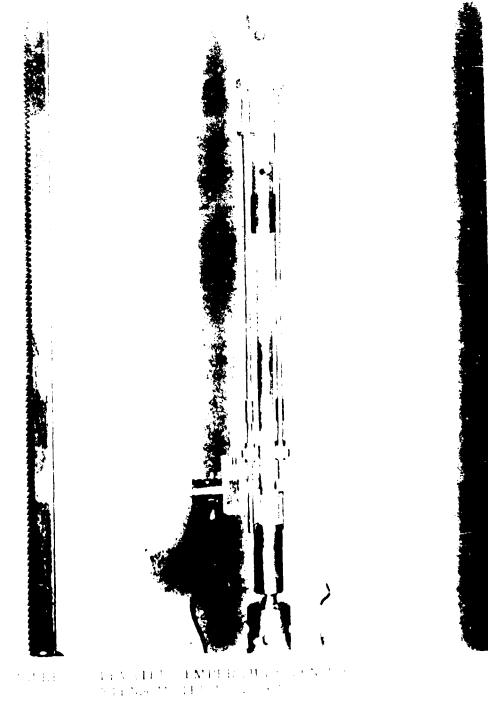


 $\frac{1}{2} \ln k \cdot \frac{1}{2} + k \cdot \frac{1}{2} \ln k \cdot$ 

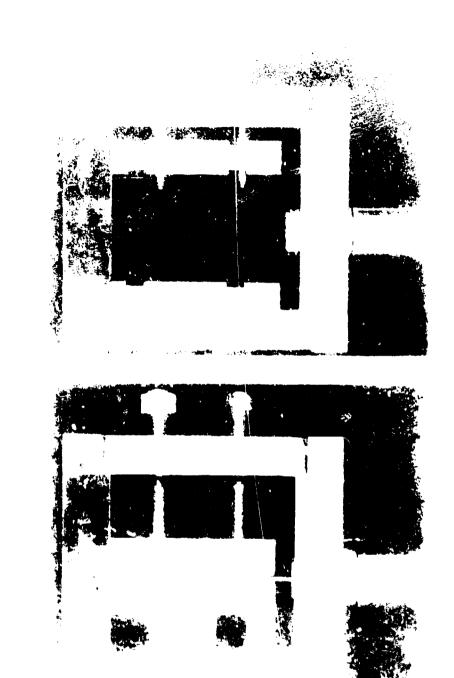






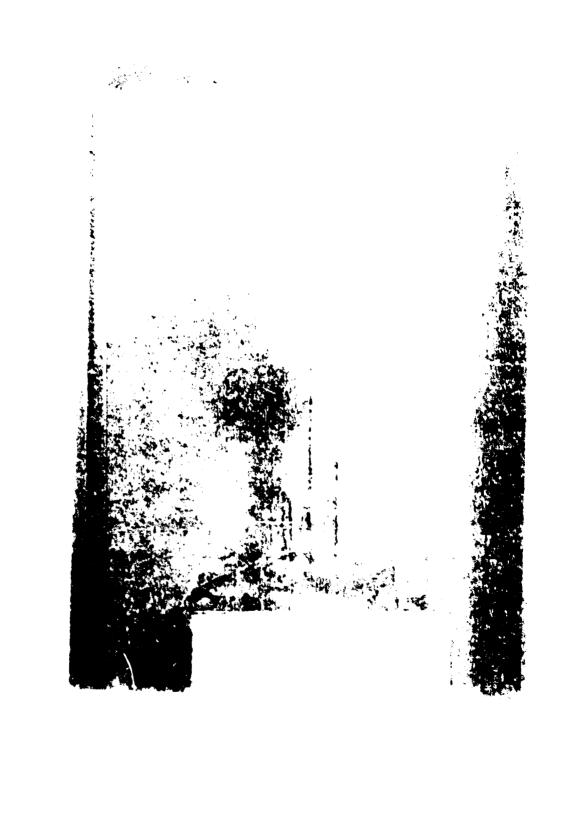


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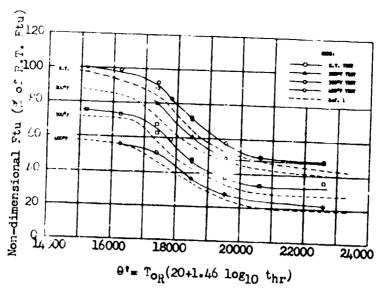
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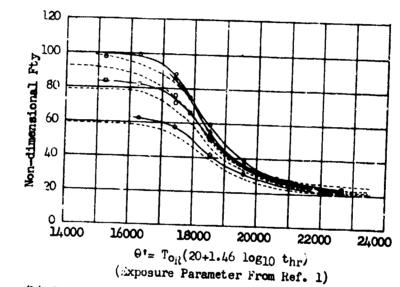
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g	78.5		78.5 78.8		78.1	70.8
5	78		85.86		78	38
			977777			Z
ቷ	69.1		7-69 7-69		7-89	
			9			
пŧ		4-7		76.7		78.9
_		7-4-				77777
						HHHH
PL		~·.		65.8 67.5		0.99
Þi		67.7		5,6		33.3
بر	بر ۾		6.6		64	
u,	79.1		78.9 78.9		71-3	
"						
					1	İ
ğ	<b>68.8</b> 69.2		68.3 69.4		67-5 67-9	
			9.9		4.0	
		<b>80</b> V		2.5		N. W.
E,		77.8		71.0	:	80.5 79.3
Z		69.2 69.5		6 <b>6.</b> 1 65.3		70°0 70°1
H		\$\$		8.2		88
بر	20		to to		2	
UIF	78.3		20,00		33	
H	##		7		7.5 7.5 7.6	
Å	67.5		68.2 68.2		67.6 67.7	
	<b>W</b> O			42		~ ~
ğ	%.5 %.0 %.0			77.5		67.5 77.5
						四月費
						L E B
Ę	68.7			6e.0 67.5		4 2
M	3%			3.5		67.2

		Fty	Ft
Sheet A	Sheet A Maximum	70,800 ps1	80,500 psi
(Above)	Minimum	65,300	76,700
	Average	68,300	78,300
Approx. universal average 67,000	d average	000,79	77,000
for 0.063 in.707	5-T6 AJCL	<b>1</b> 00	

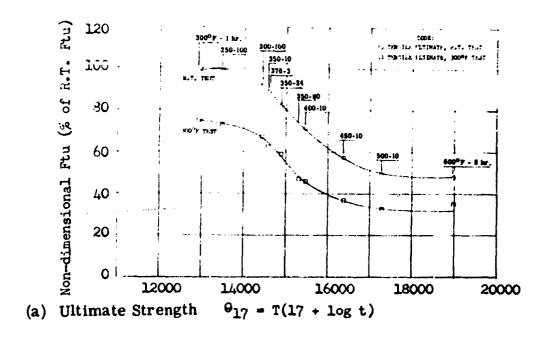


## (a) Ultimate Strength



## (b) Yield Strength

FIGURE 12 COMPARISON OF STRENGTH AFTER UNSTRESSED EXPOSURE FOR TWO LOTS OF 7075-T6



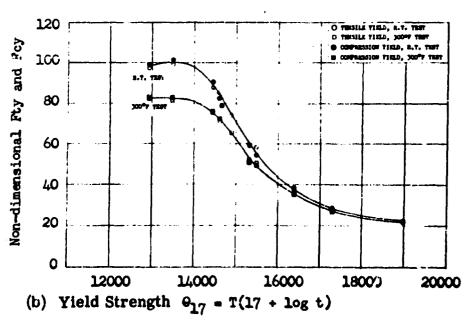
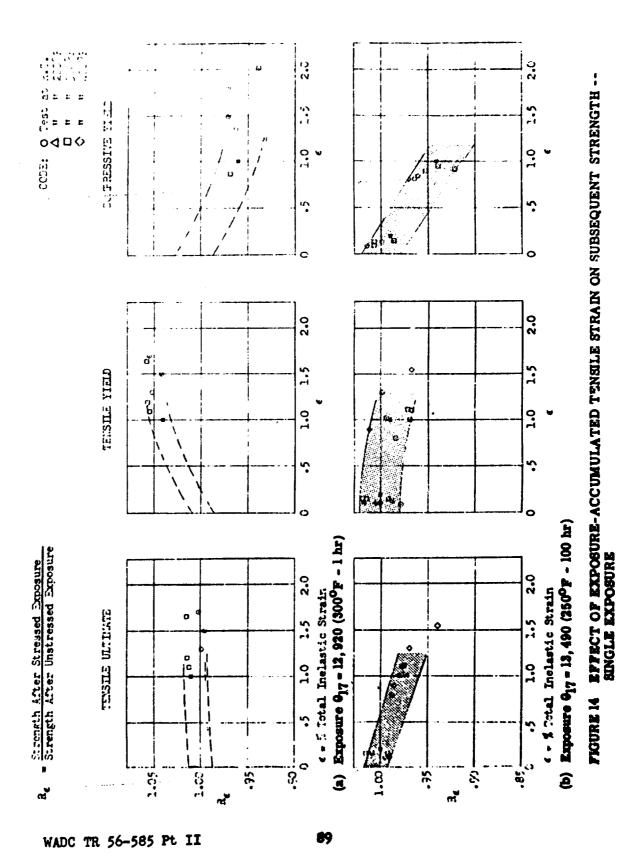


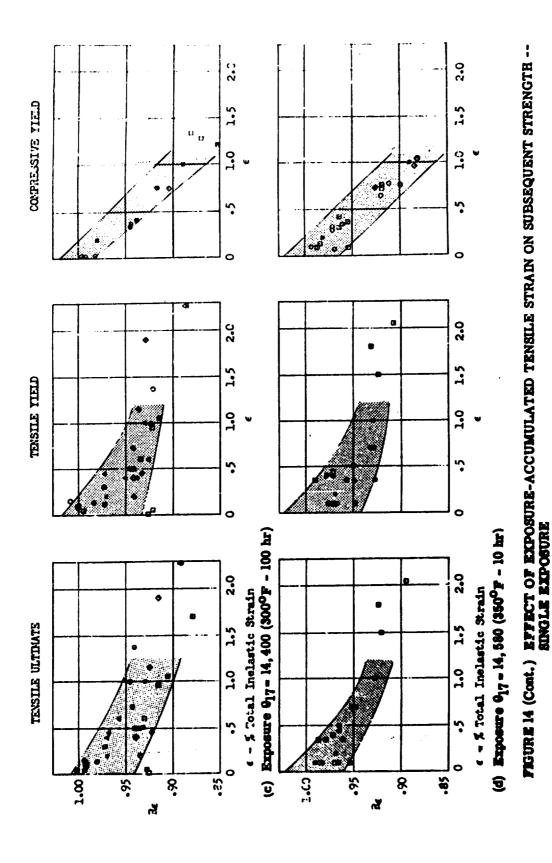
FIGURE 13 CORRELATION OF STRENGTH REDUCTION AFTER EXPOSURE WITH EXPOSURE PARAMETER 917

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Best Available Copy



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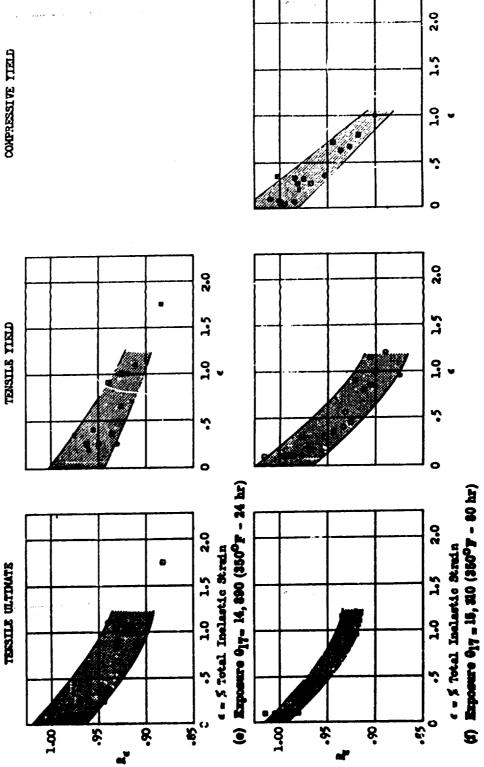
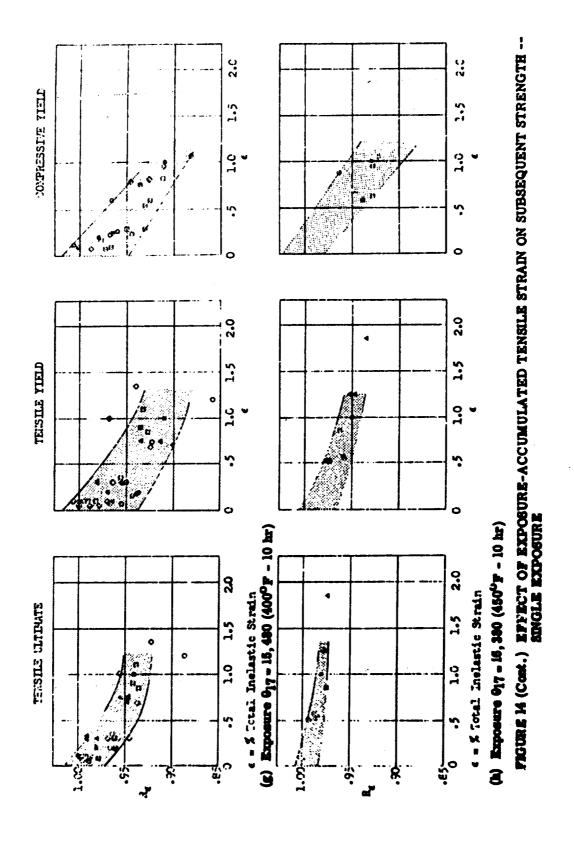


FIGURE 14 (Cont.) EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTH ---



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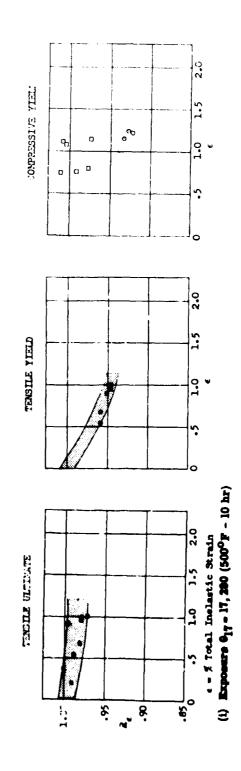
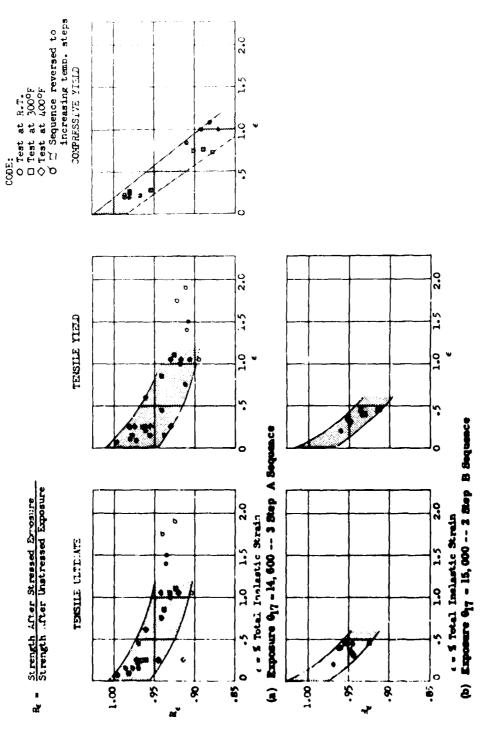


FIGURE 14 (COST.) EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTH ---



PIGURE 15 EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTH -MULTIPLE EXPOSURE

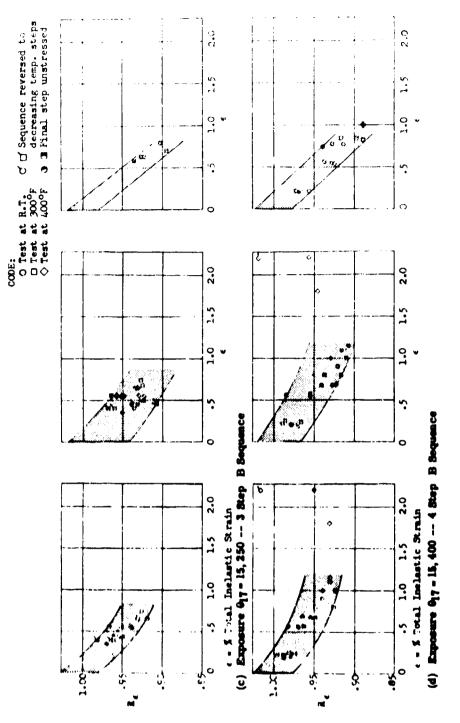
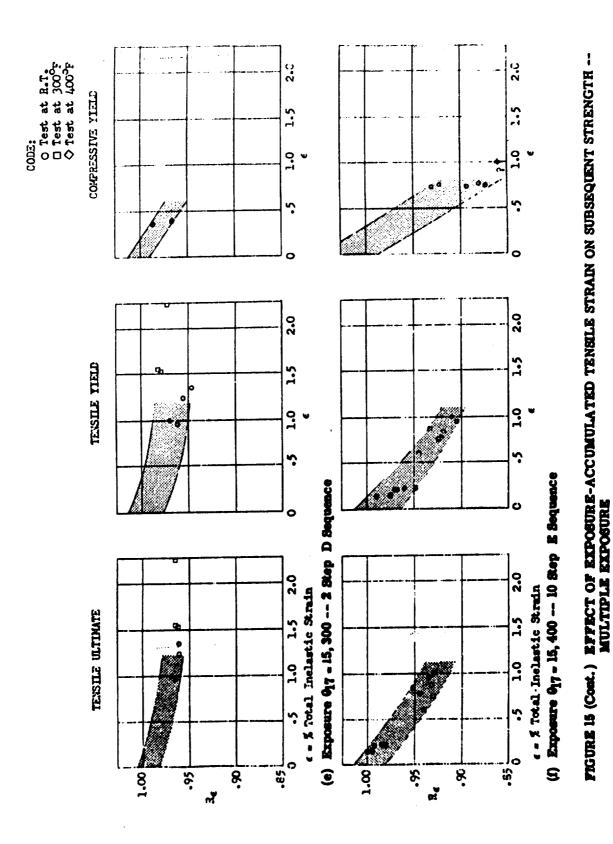


FIGURE 15 (COM.) EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTH ---

A. . . .



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- Single Exposure
  Hultiple Exposure
- n. Multiple Exposure (D Sequence, 1 % inelastic strain)

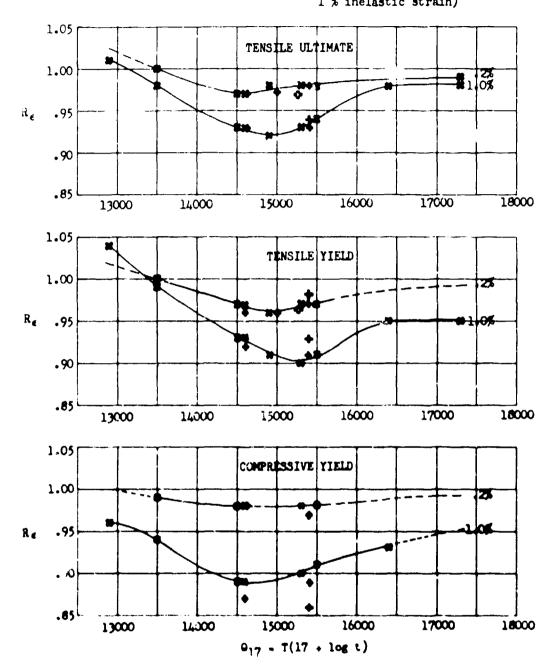


FIGURE 16 VARIATION OF STRESSED EXPOSURE EFFECTS WITH DEGREE OF EXPOSURE

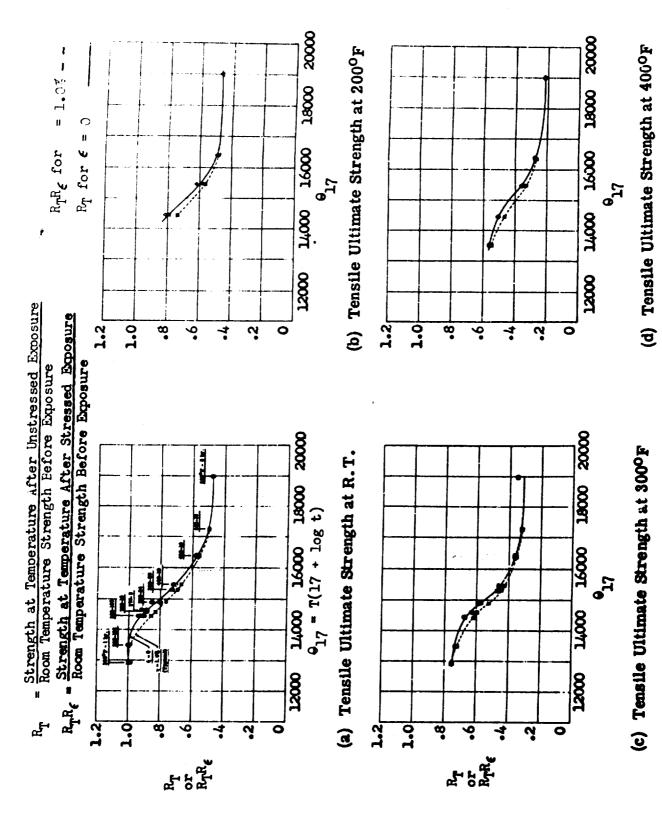
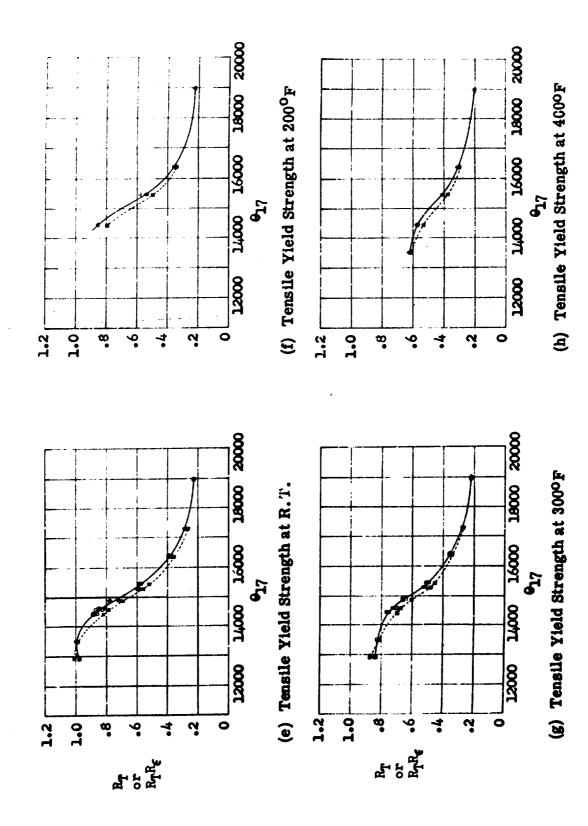
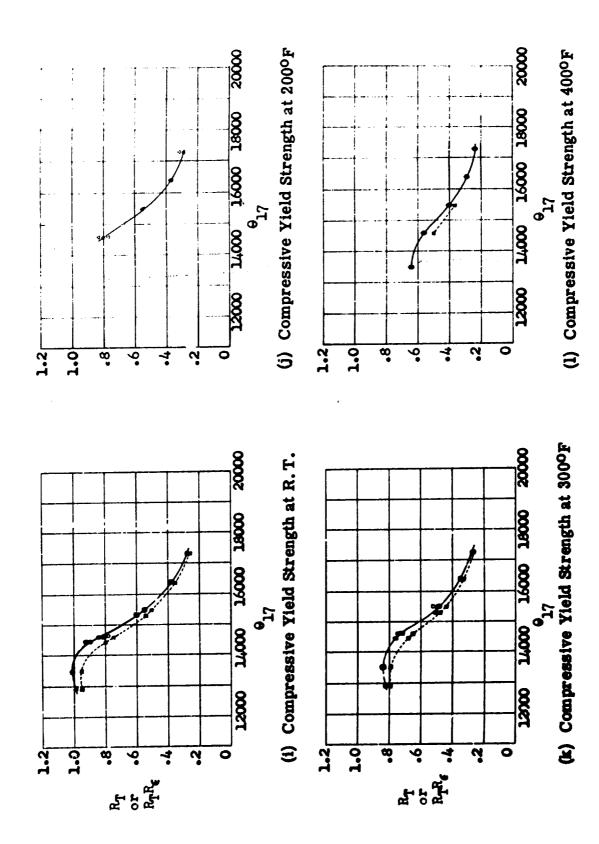
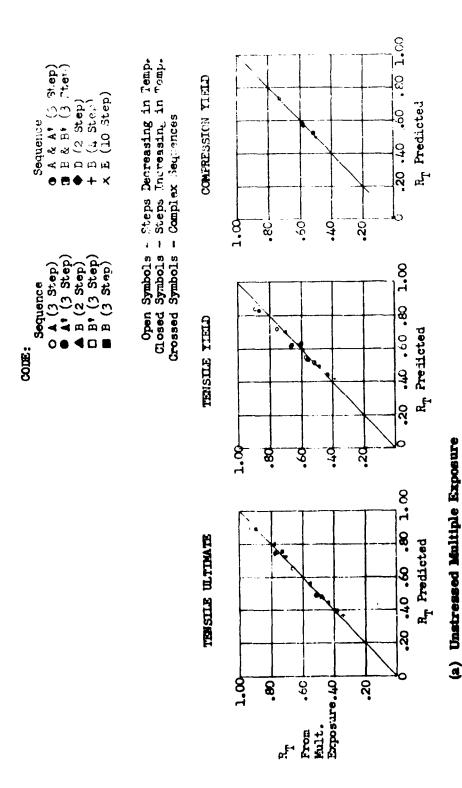


FIGURE 17 STRENGTH REDUCTION AFTER UNSTRESSED AND STRESSED SINGLE EXPOSURE





100



COMPARISON OF STRENGTH AFTER MULTIPLE EXPOSURE WITH STRENGTH PREDICTED FROM SINGLE EXPOSURE CURVES FIGURE 18

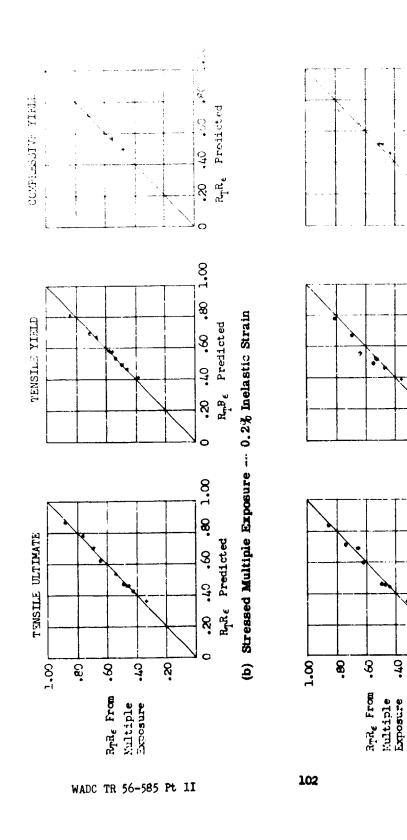


FIGURE 18 (Cont.) COMPARISON OF STRENGTH AFTER MULTIPLE EXPOSURE WITH STRENGTH PREDICTED FROM SINGLE EXPOSURE CURVES

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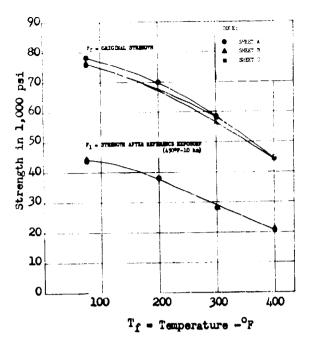
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RyRe Predicted

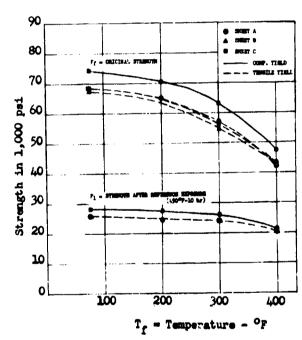
MrR Predicted

(c) Stressed Multiple Exposure -- 1.0% Inclustic Strain

RyR<sub>E</sub> Predicted

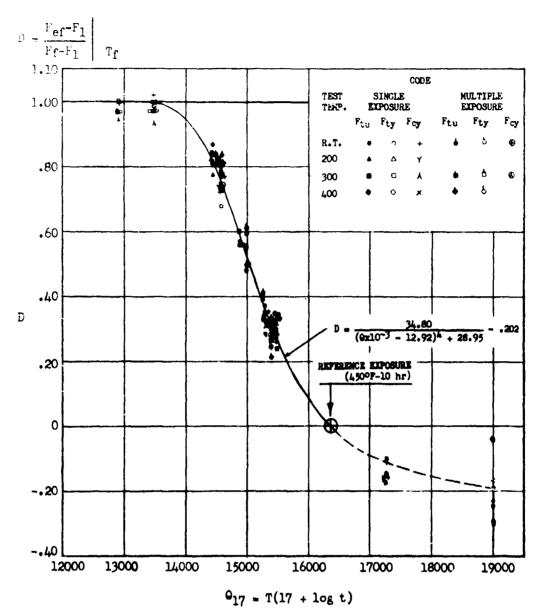


(a) Ultimate Tensile Strength



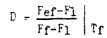
(b) Yield Strength

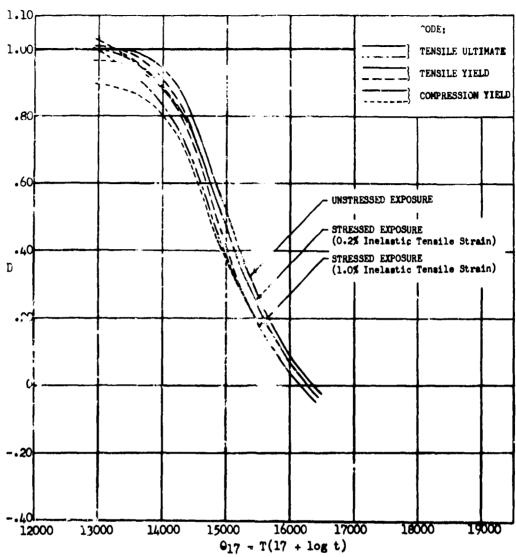
FIGURE 19 STRENGTH OF 7075-T6 AT TEMPERATURE BEFORE AND AFTER REFERENCE EXPOSURE



## (a) Unstressed Exposures

FIGURE 20 STRENGTH DETERIORATION CHARACTERISTICS OF 7075-T6 AT ALL TEST TEMPERATURES





## (b) Stressed Exposures

FIGURE 20 (Cont.) STRENGTH DETERIORATION CHARACTERISTICS OF 7075-T6 AT ALL TEST TEMPERATURES

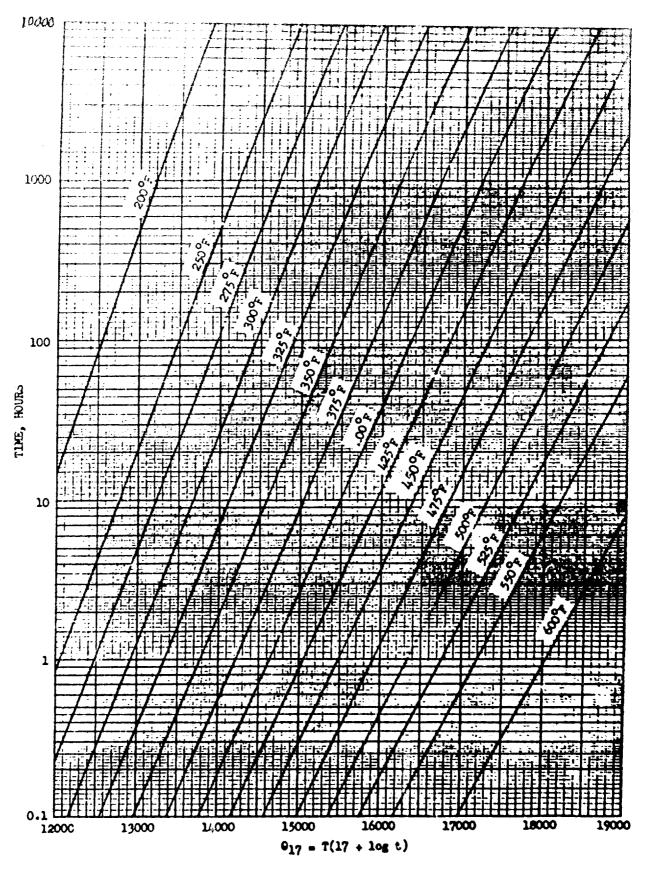


FIGURE 21 LARSON-MILLER PARAMETER DIAGRAM
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